

物構研シンポジウム'09

放射光・中性子・ミュオンを用いた表面・界面科学の最前線

2009年11月17-18日(17日(金))@つくば国際会議場エポカル

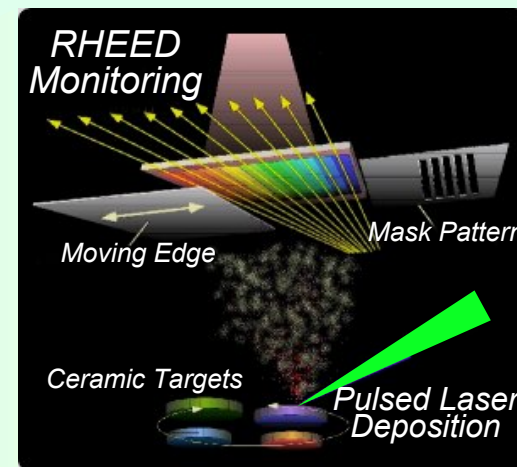
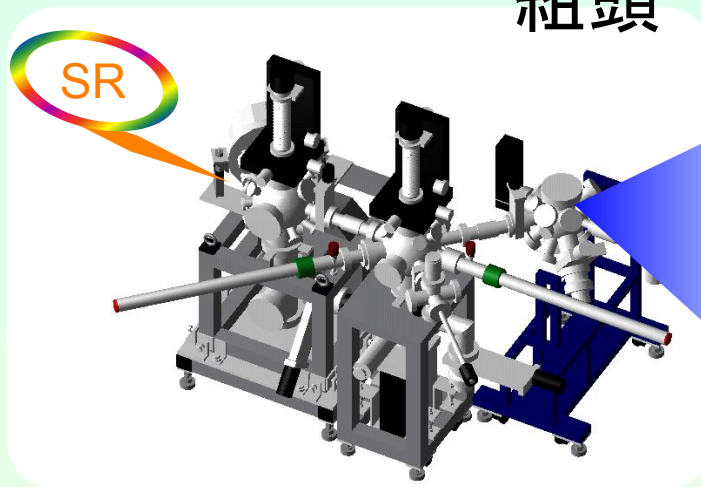
その場放射光電子分光による 強相関ヘテロ界面の電子状態の研究

東京大学大学院工学系研究科

JST-さきがけ

東大放射光連携研究機構

組頭 広志



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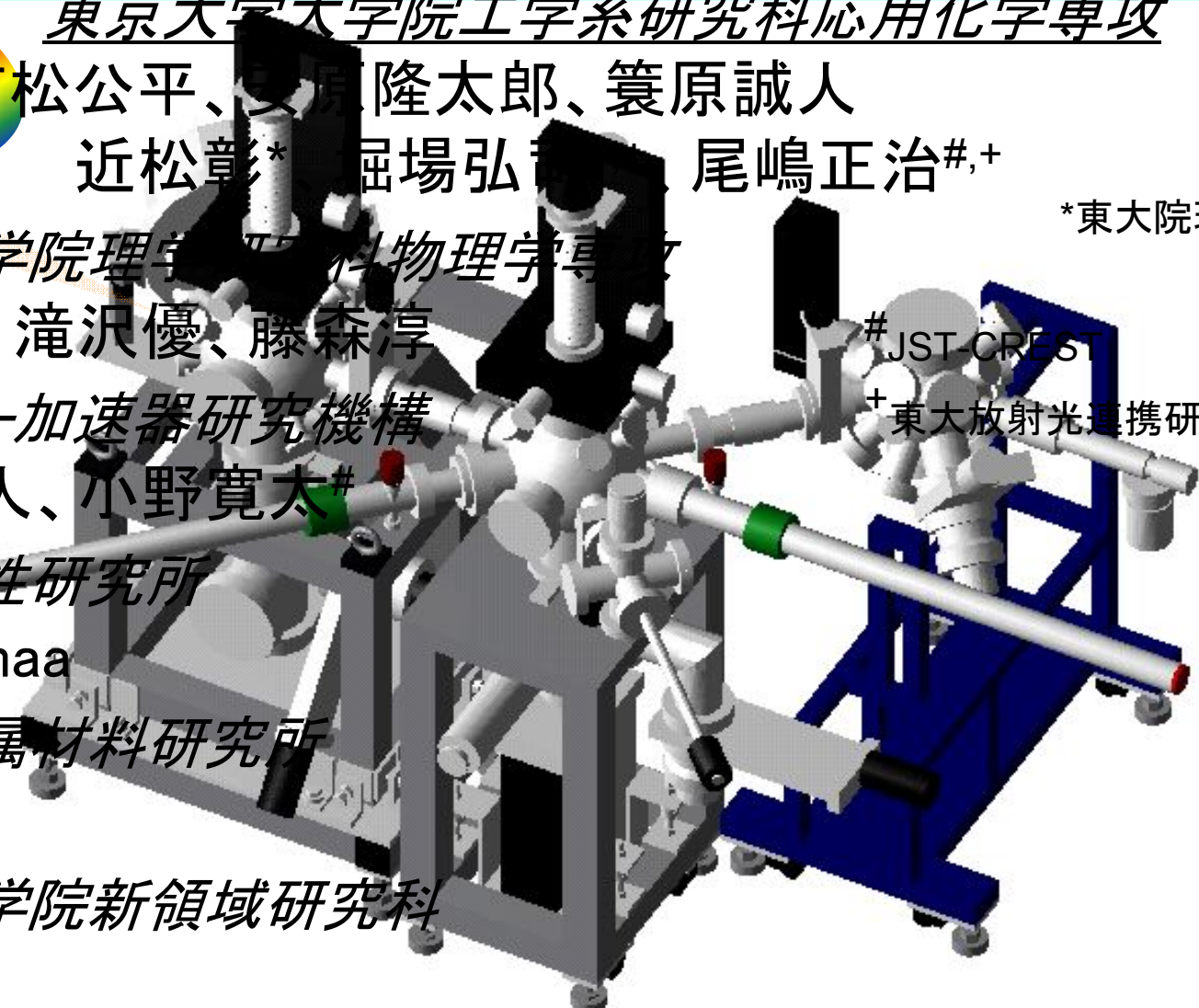
川崎雅司#

東京大学大学院新領域研究科

鯉沼秀臣

#JST-CREST

+東大放射光連携研究機構



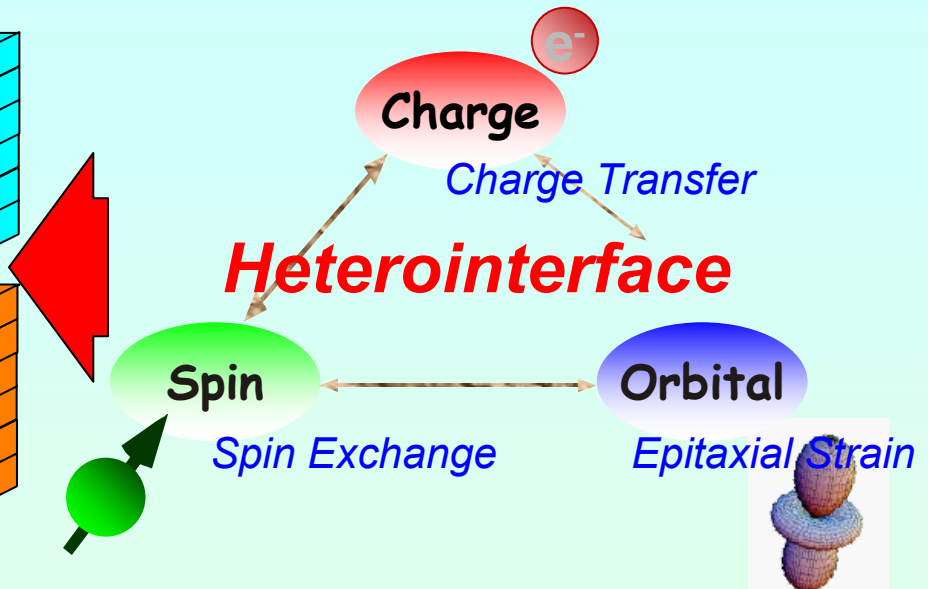
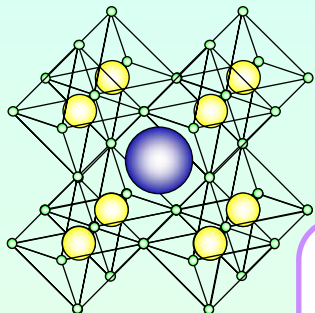
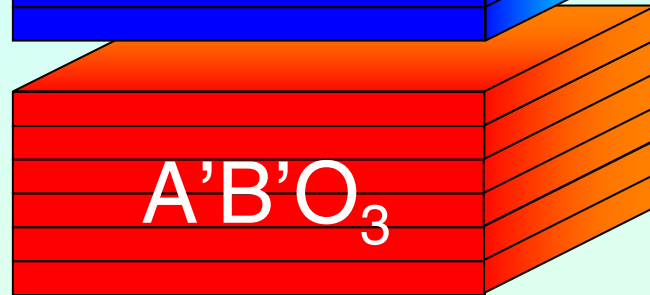
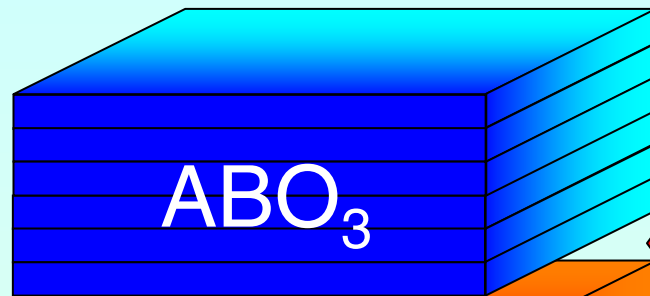
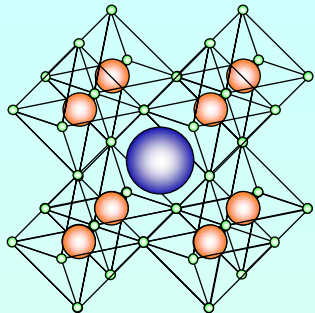
はじめに

1. 成膜技術（レーザーMBE法など）の発展により、原子レベルで構造を制御した酸化物薄膜・超格子の作製が可能となった。
2. 強相関酸化物においても、今後、物質設計の自由度が高く、物性を自由自在に制御可能な「超構造」を用いた研究が主流になる。
3. 放射光を用いた分光法は、酸化物薄膜・超格子研究の強力な牽引力になる。

INTRODUCTION

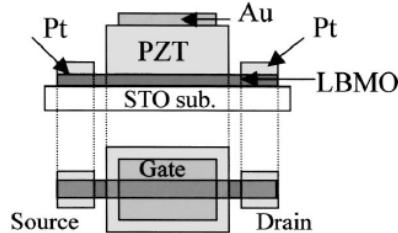
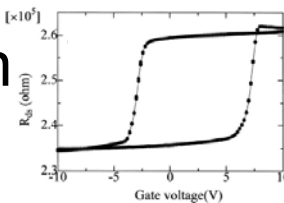
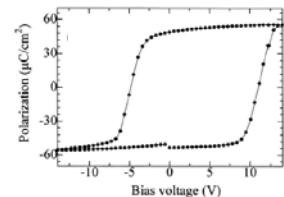
Heterojunctions based on Perovskite Oxides

Controlling physical properties using interface effects



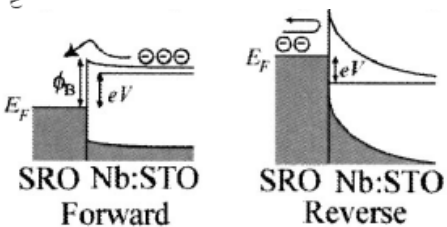
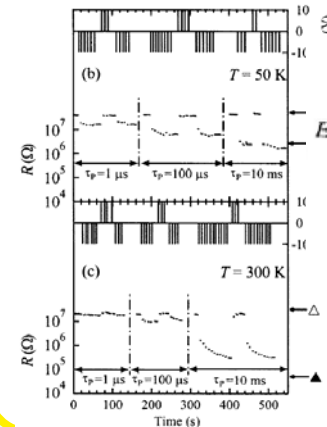
High Tc
CMR
MI transition

Ferroelectric FET



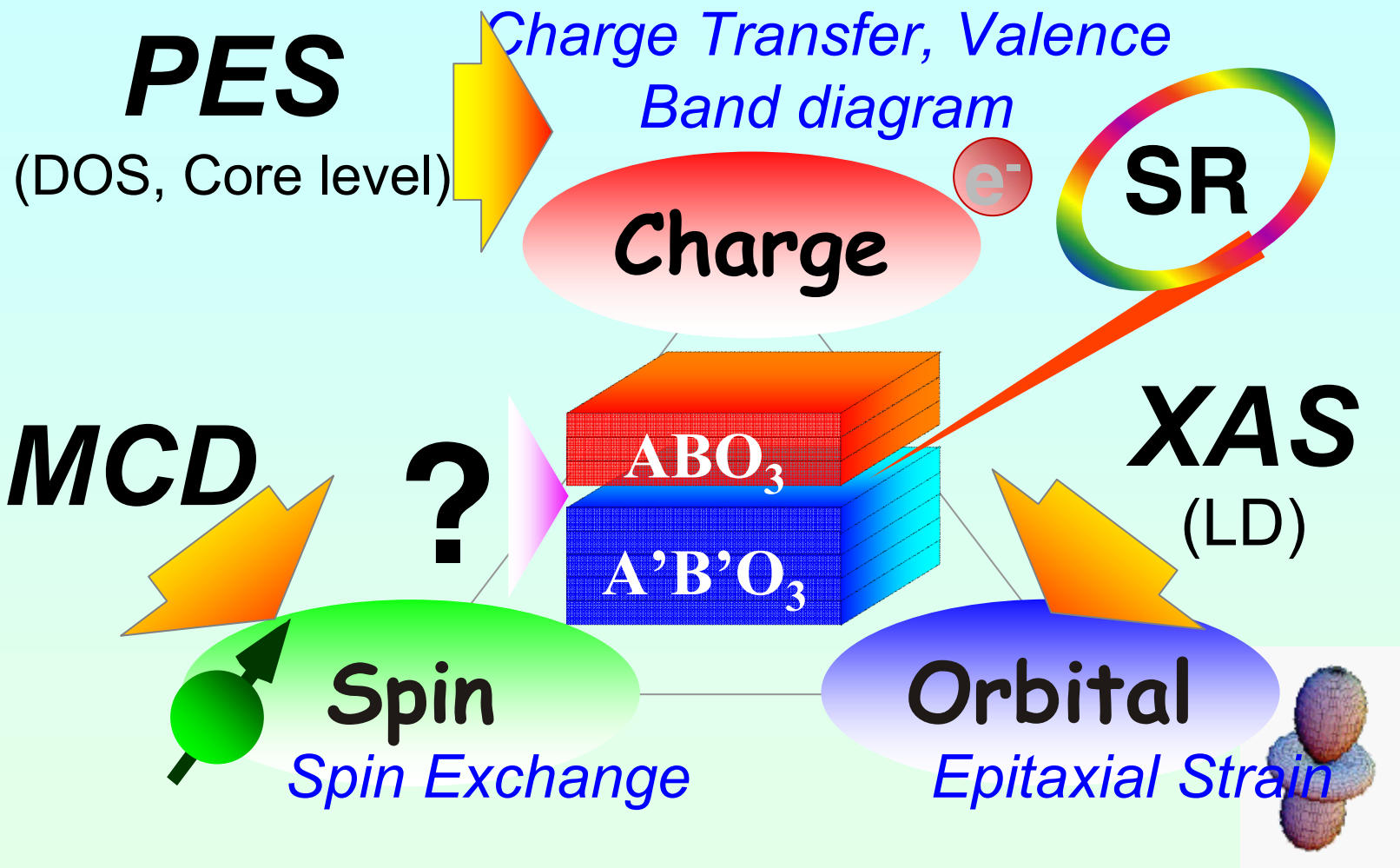
T. Kanki *et al.*, *Appl. Phys. Lett.* **83**, 4860 (2003).

Resistance RAM



T. Fujii *et al.*, *Appl. Phys. Lett.* **86**, 012107 (2005).

SR analysis for oxide heterointerface



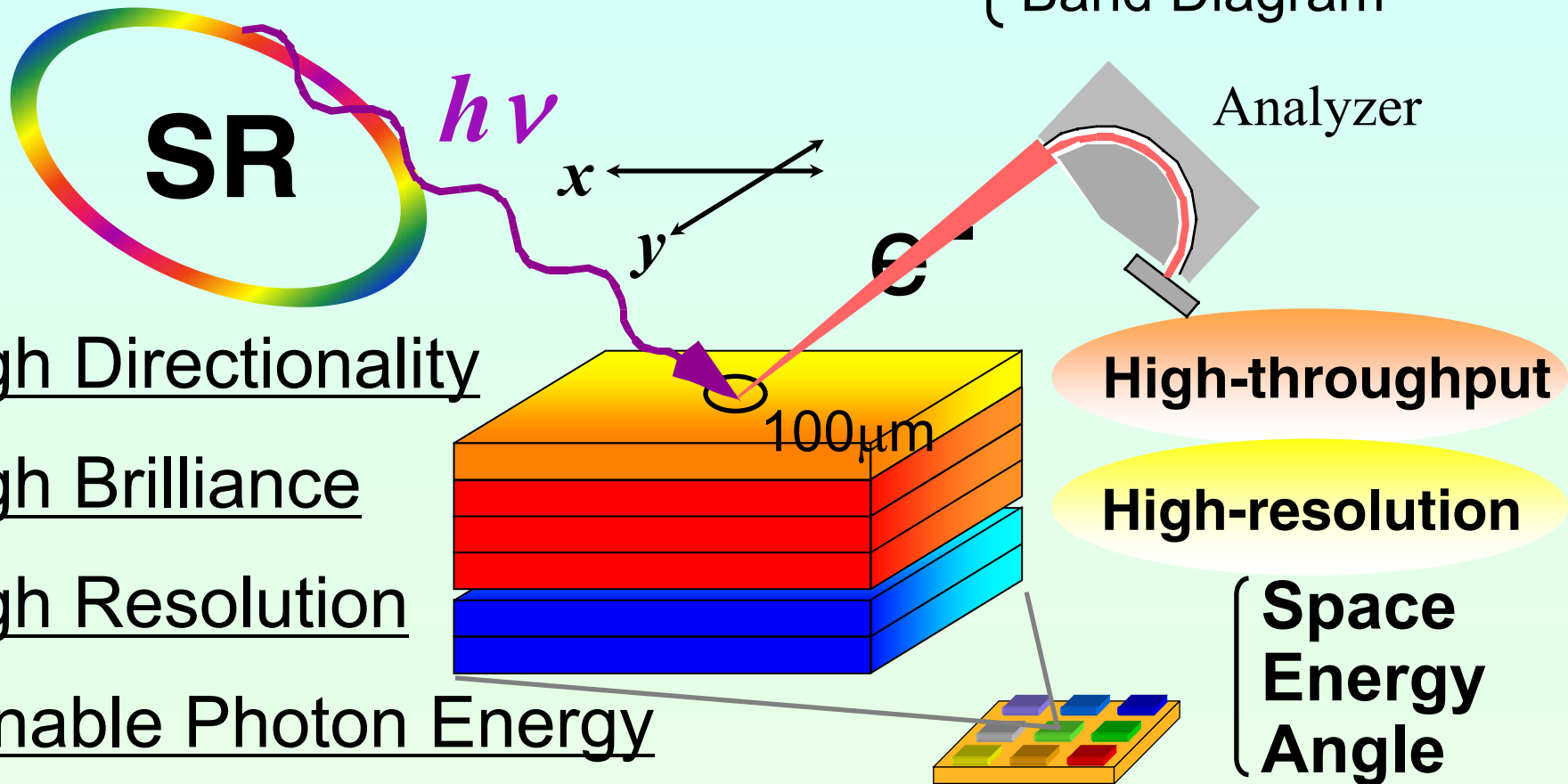
界面数nmの領域(少ないボリューム)における
電子・スピン・軌道状態を元素選択的に測定可能

Advantage of SR-PES

- ☆ Non-destructive
- ☆ Surface (Interface) Sensitive
- ☆ Direct Determination of Electronic States

Hard-XrayPES@SPring-8 ($\sim 1000 \text{ \AA}$)

($5 \sim 30 \text{ \AA}$)
Chemical Shift
DOS
Band Diagram



High Directionality

High Brilliance

High Resolution

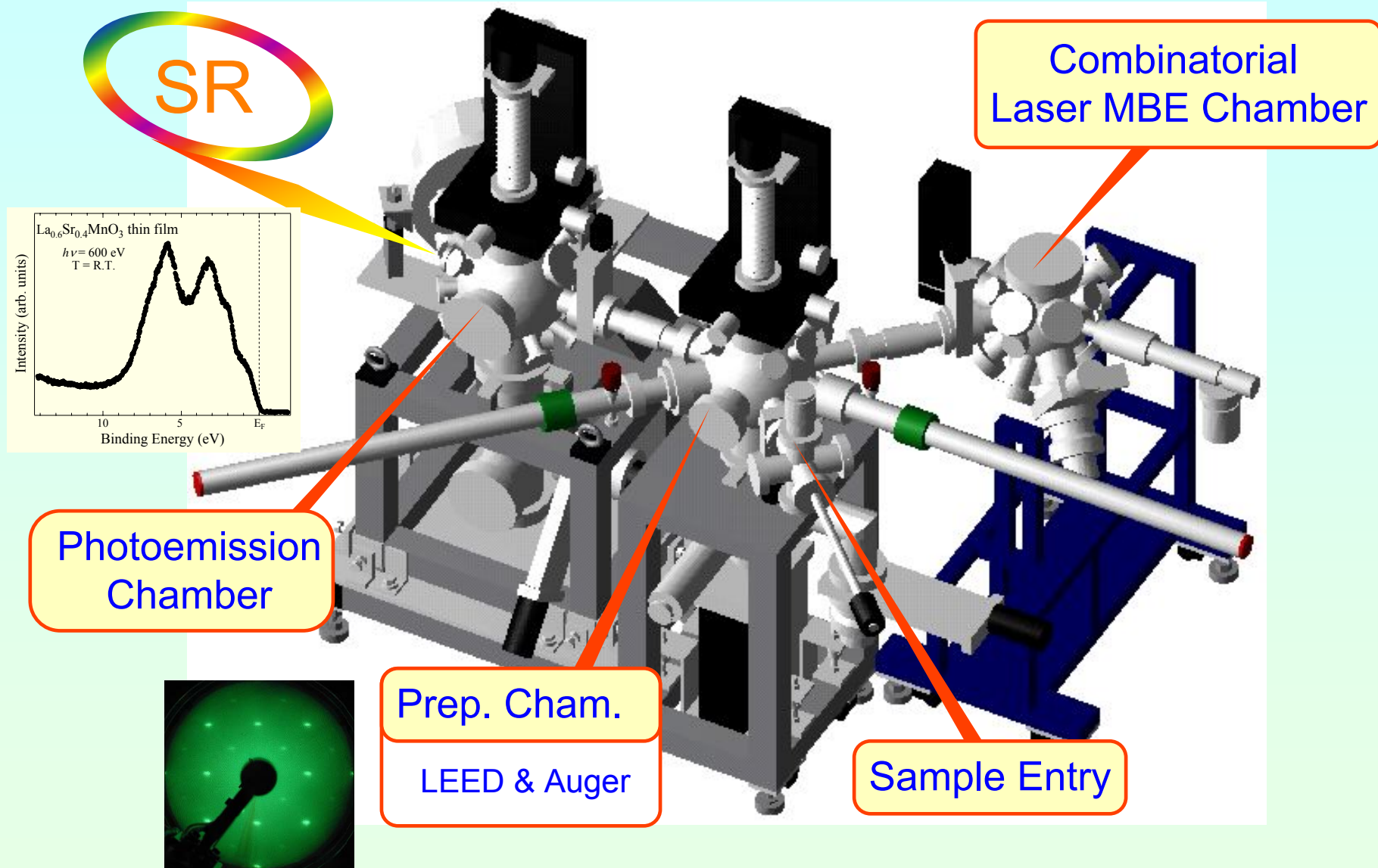
Tunable Photon Energy

High-throughput

High-resolution

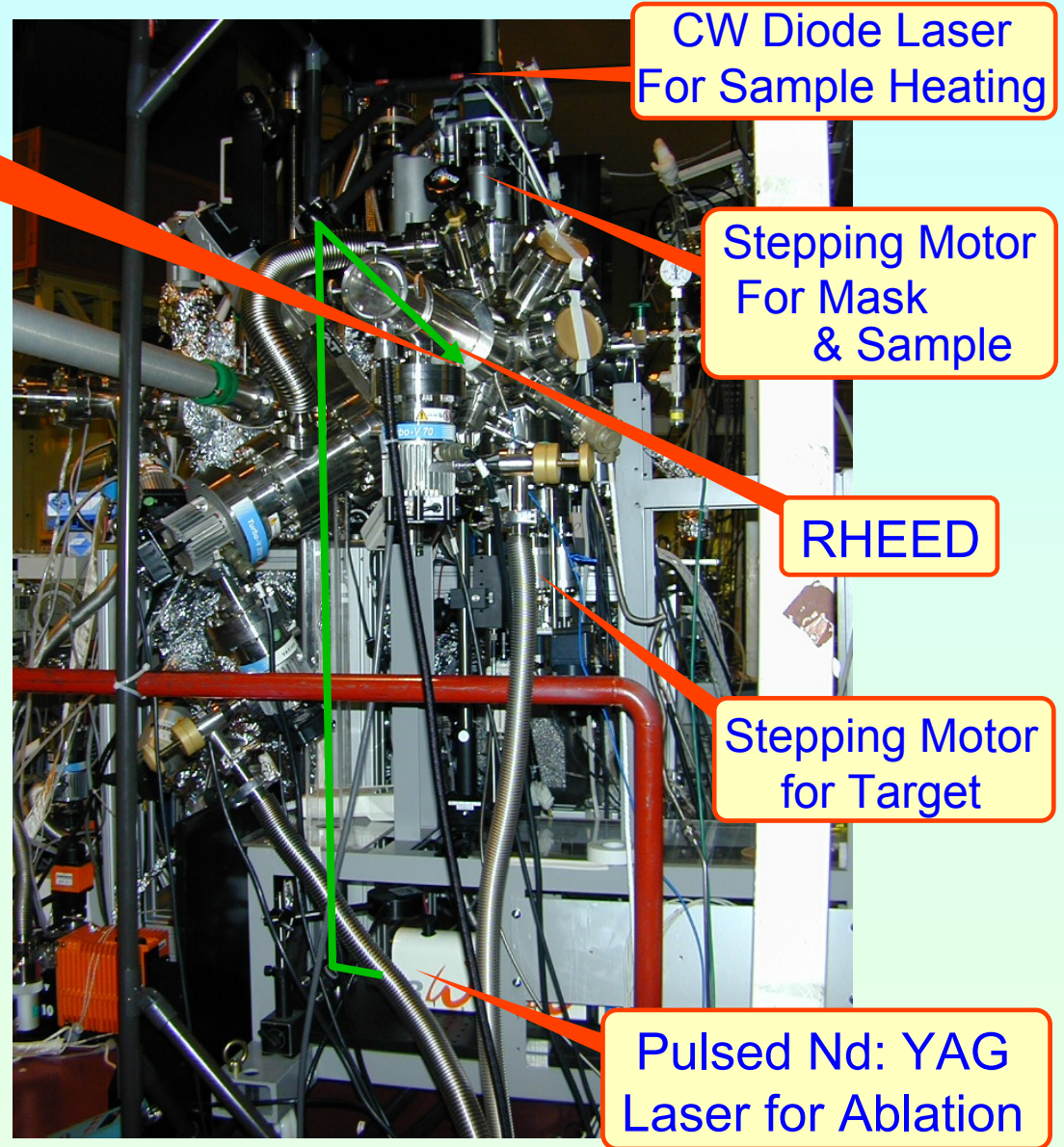
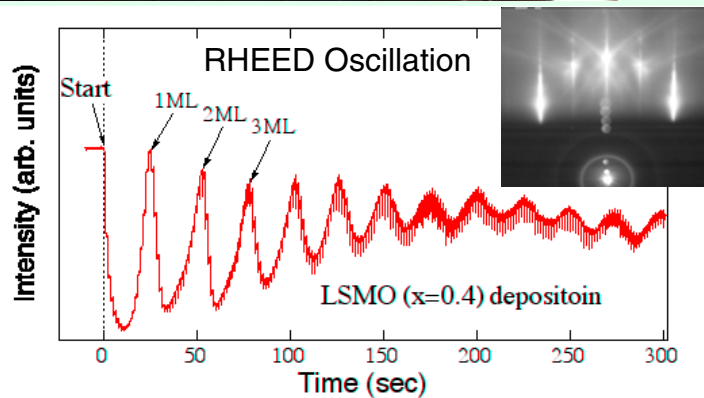
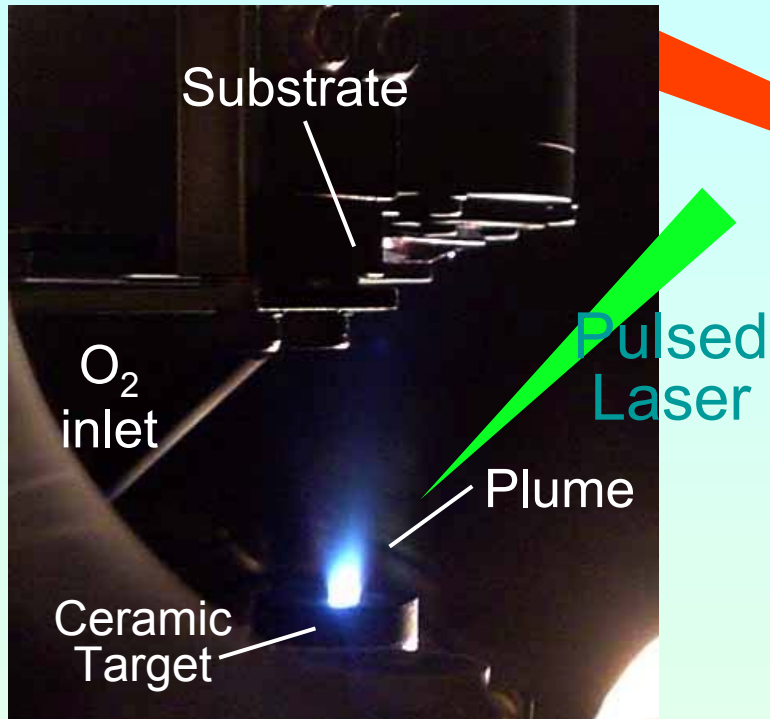
Space
Energy
Angle

In-situ PES + Laser MBE system

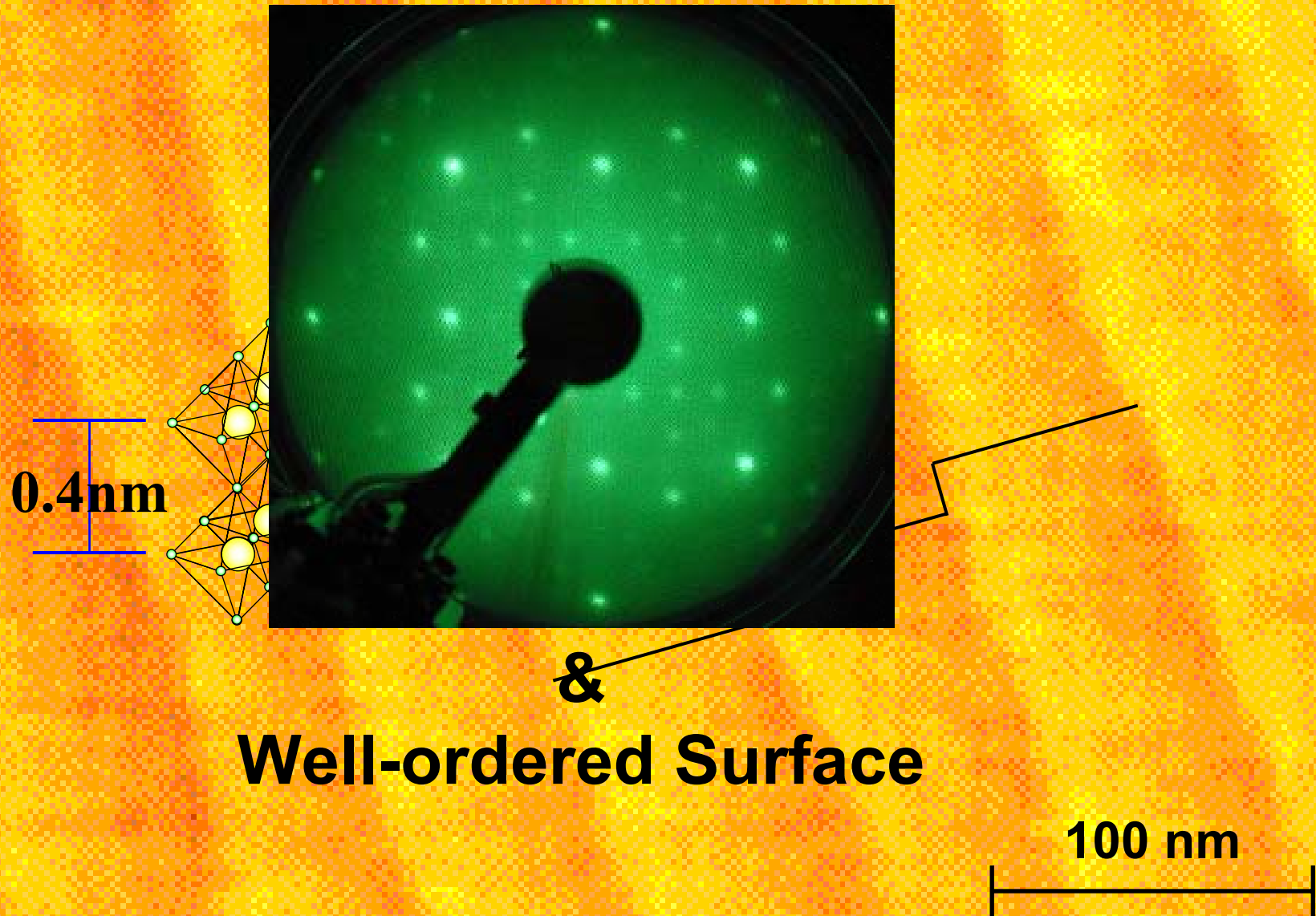


Combinatorial Laser MBE Apparatus

Laser Molecular Beam Epitaxy

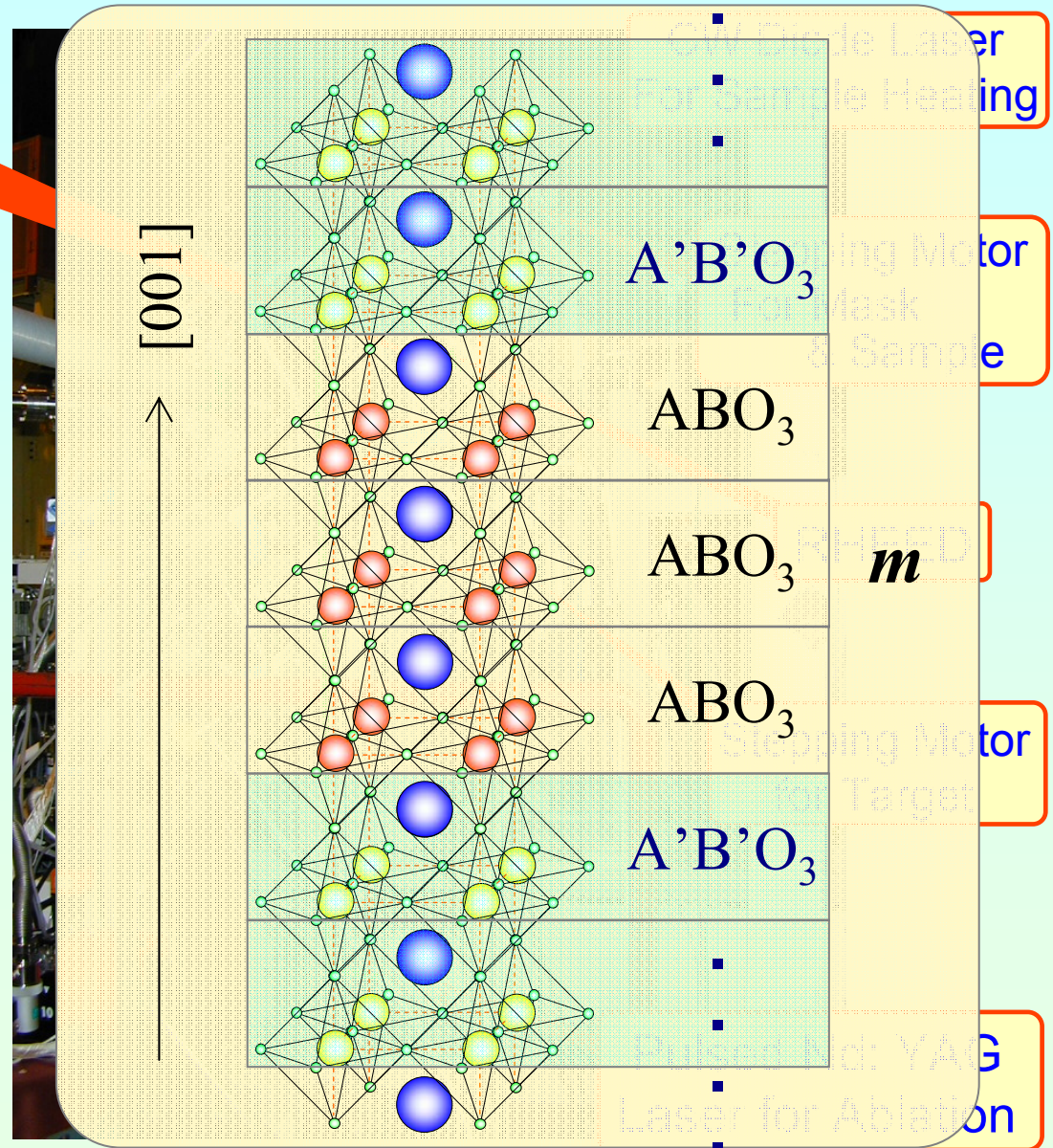
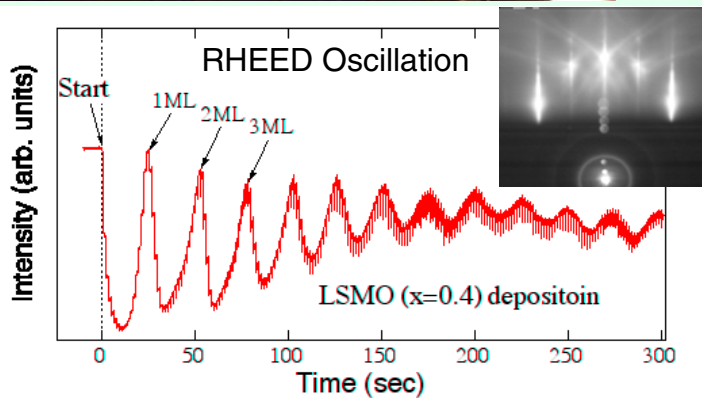
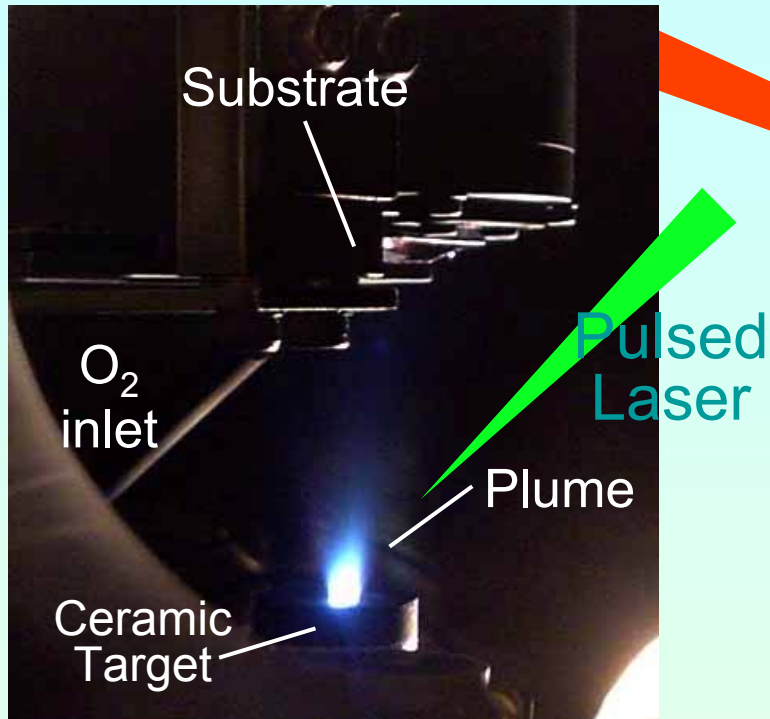


Atomically-flat surface of $\text{La}_{1-x}\text{Sr}_x\text{MnO}_3$ thin films



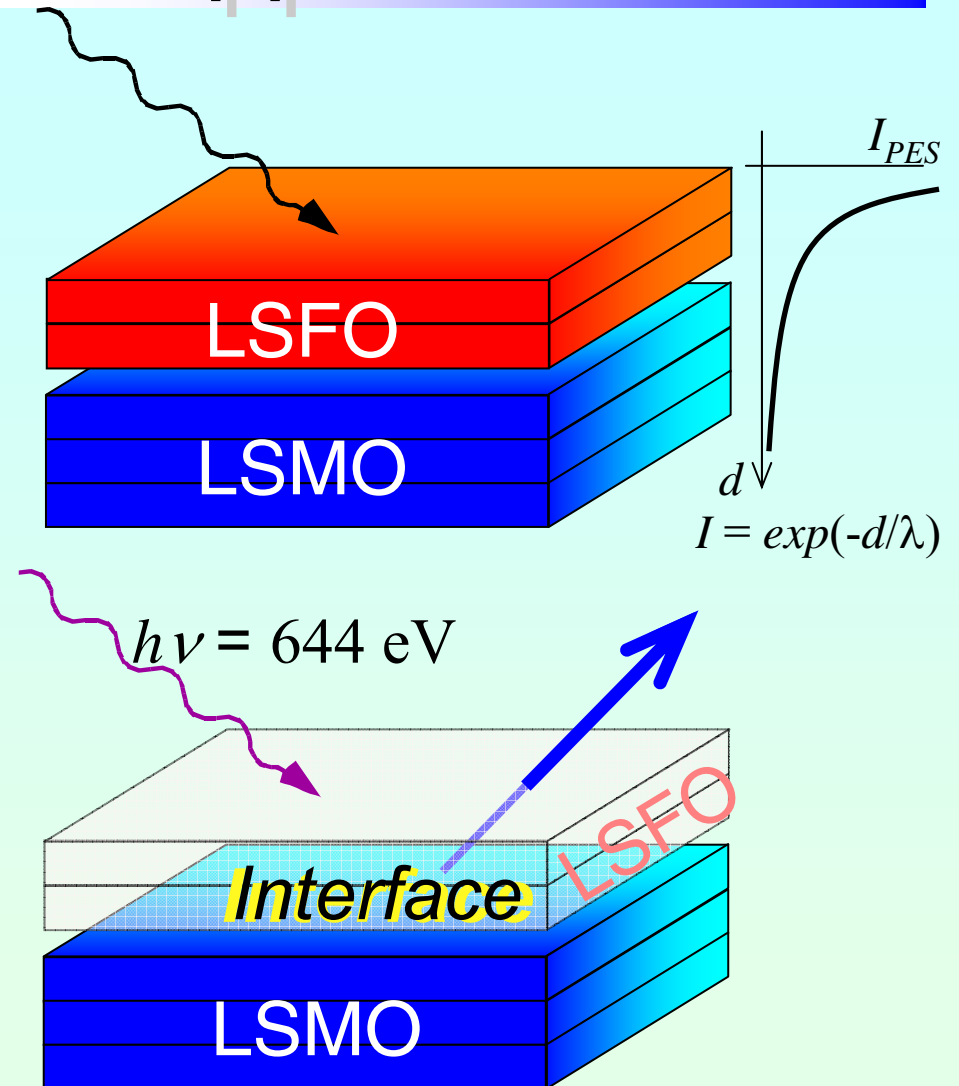
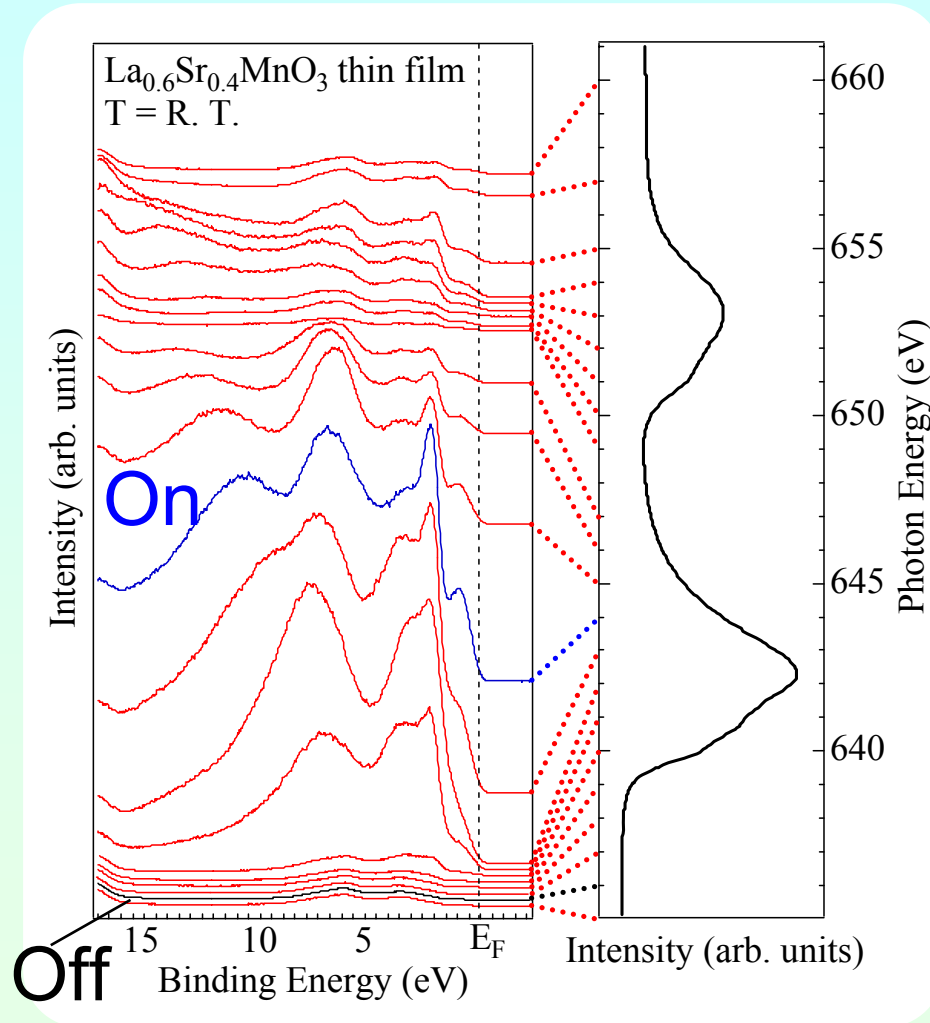
Combinatorial Laser MBE Apparatus

Laser Molecular Beam Epitaxy



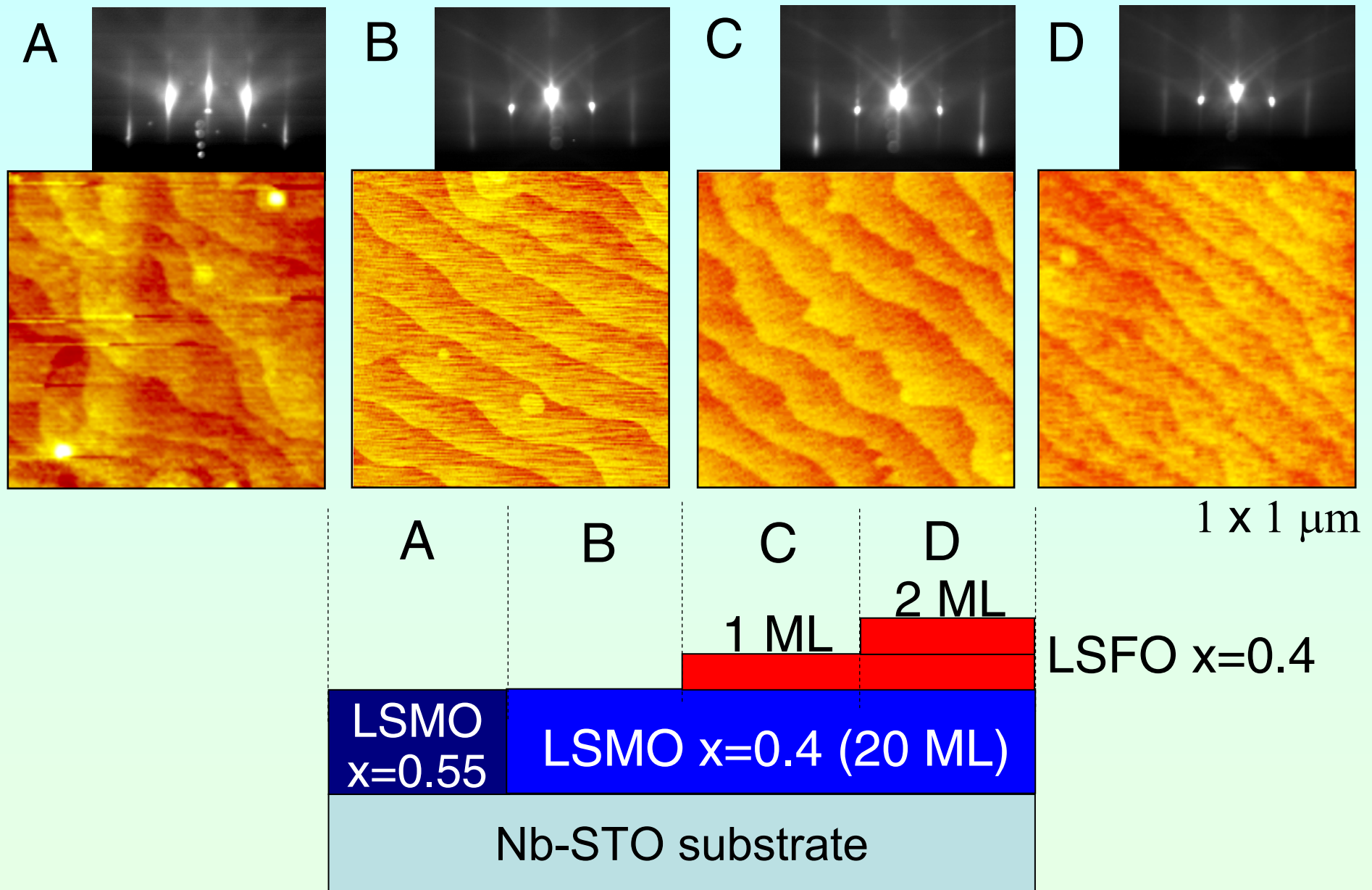
Resonant PES Approach

Resonant PES of LSMO (x=0.4)



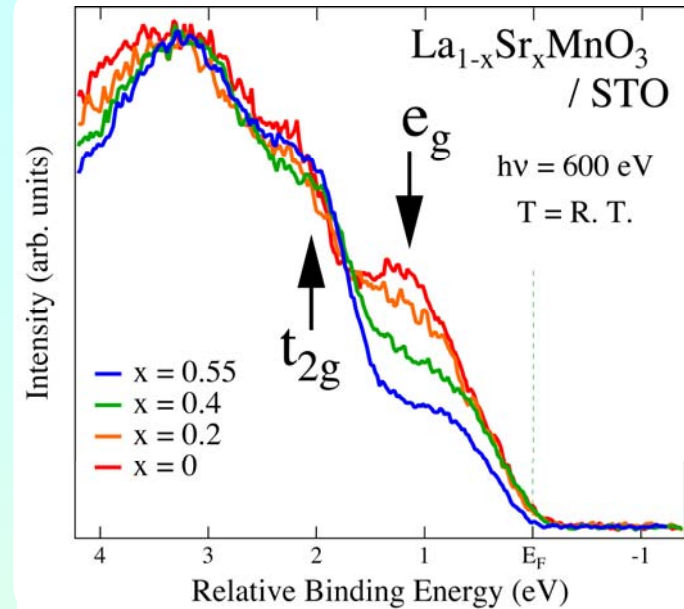
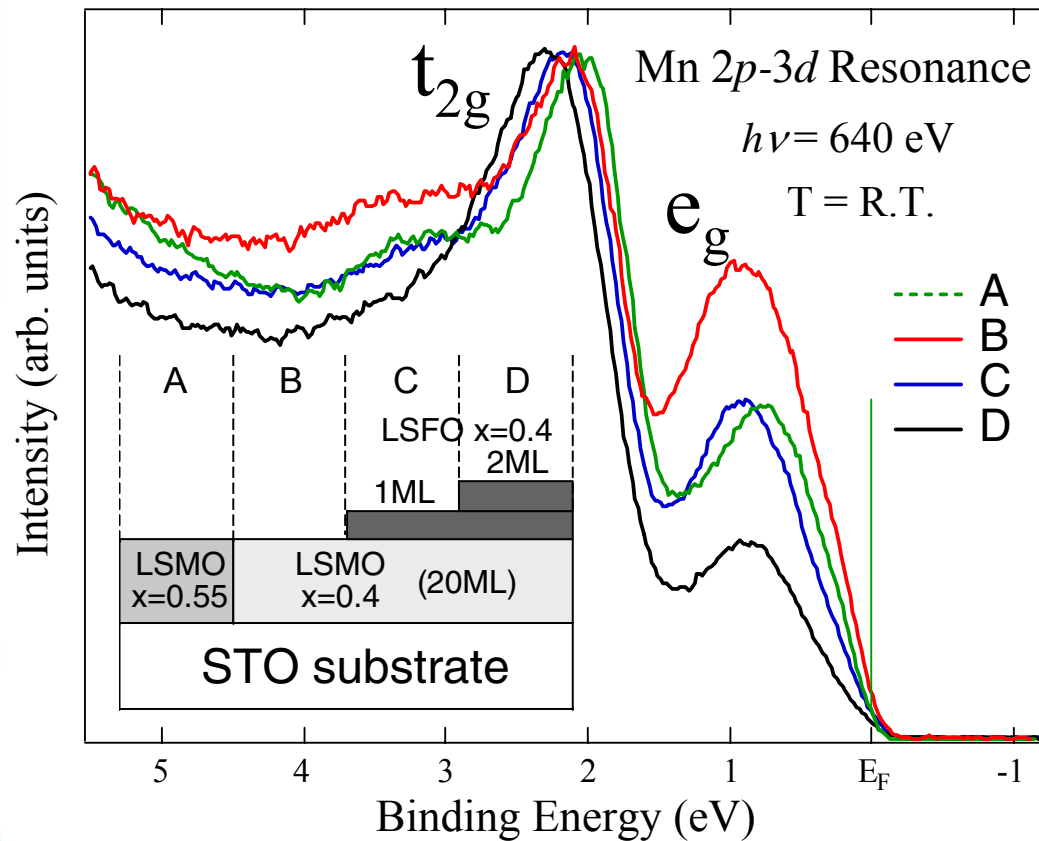
Probing Electronic Structure
at the Interface (Mn 3d PDOS)

RHEED Pattern & AFM Images

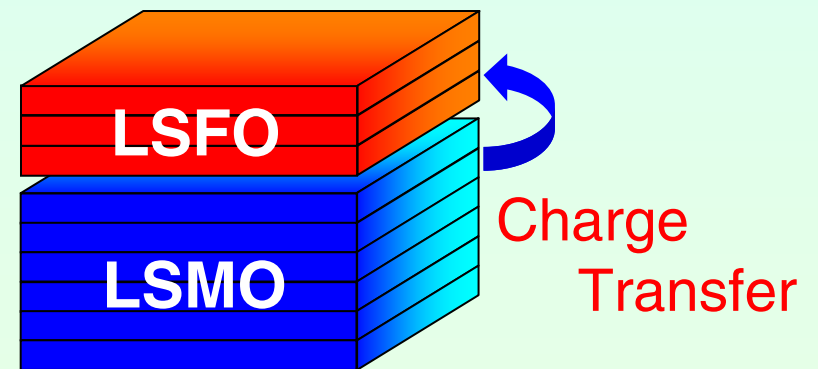


Resonant PES of LSMO at Interface

Mn 2p-3d Resonant PES



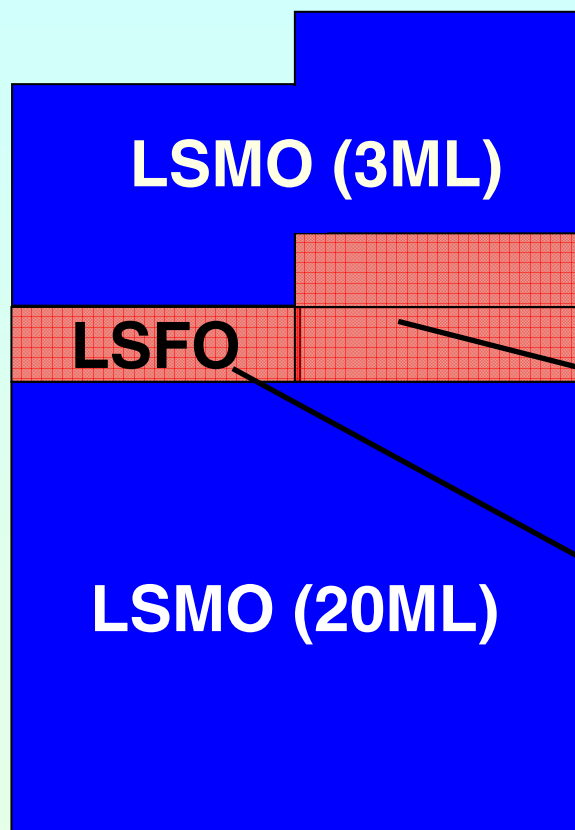
K. Horiba, H.K. *et al.*, Phys. Rev. B **71**, 155420 ('05)



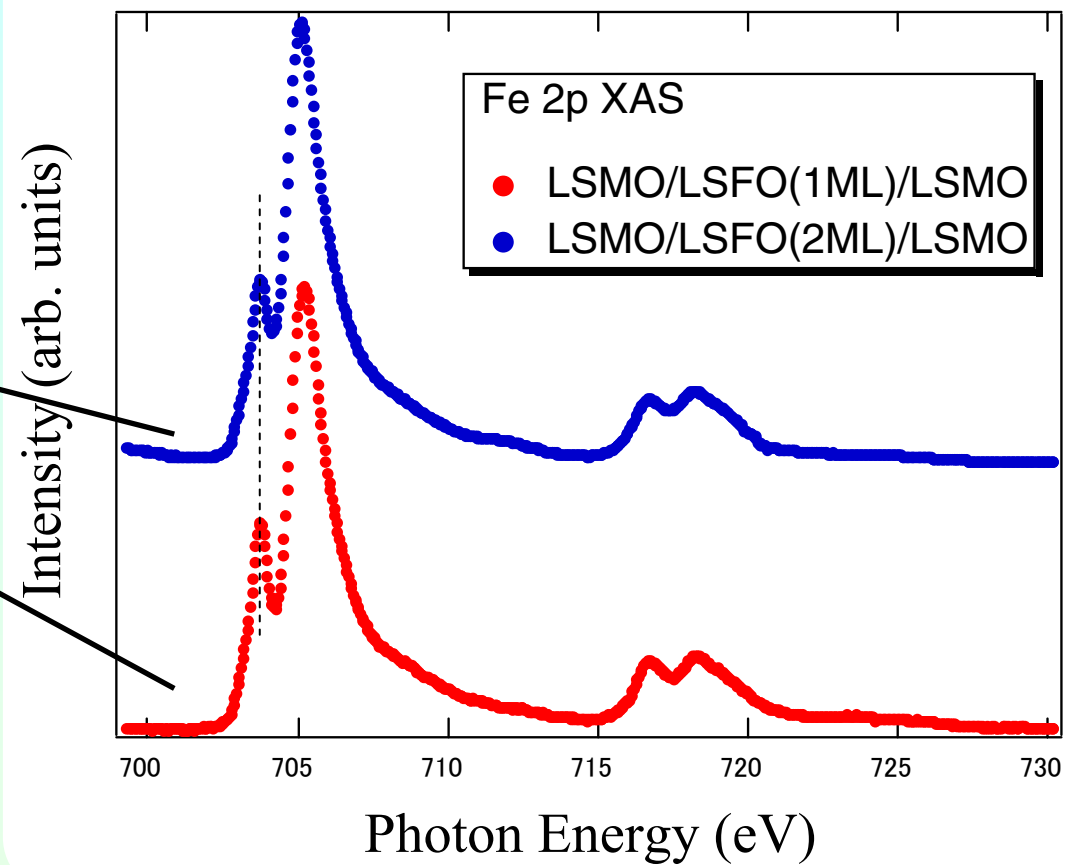
Spectral Evidence of Charge Transfer at LSMO/LSFO Interface

H. Kumigashira *et al.*, Appl. Phys. Lett. **84**, 5353 (2004).

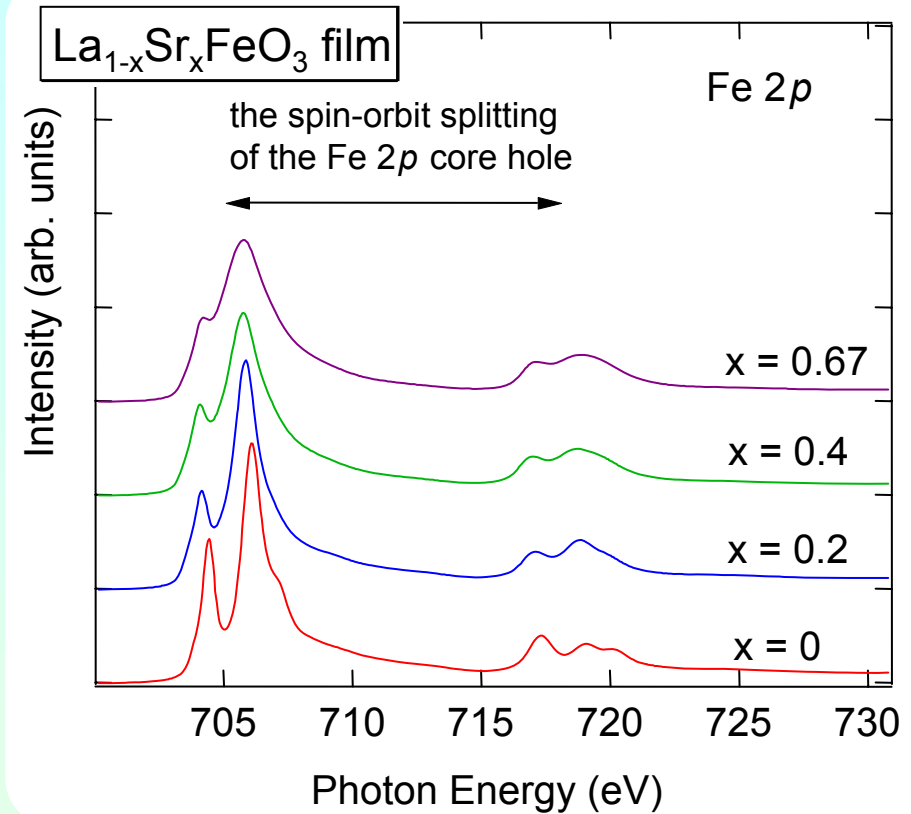
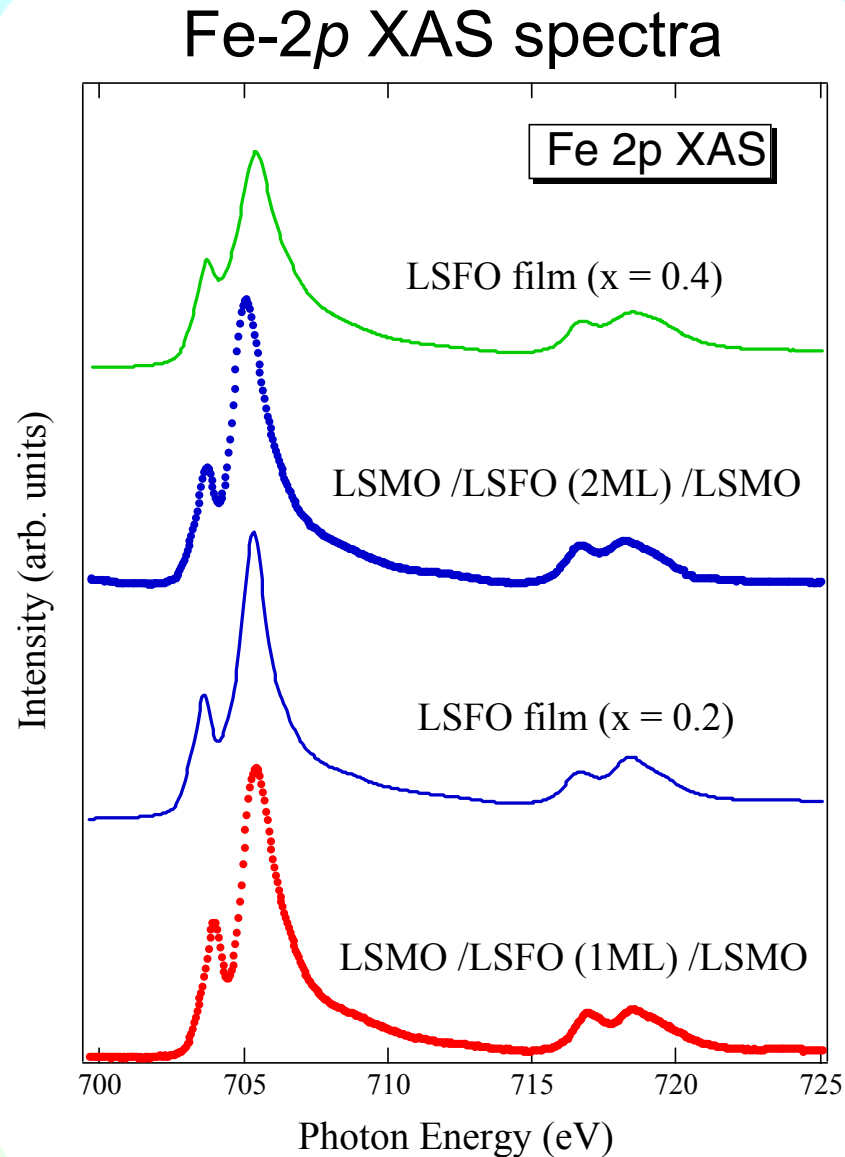
XAS Spectra of Interfacial LSFO Layer



Fe-2p XAS spectra



Comparison of Fe-2p XAS spectra between Interfacial LSFO layer and LSFO films

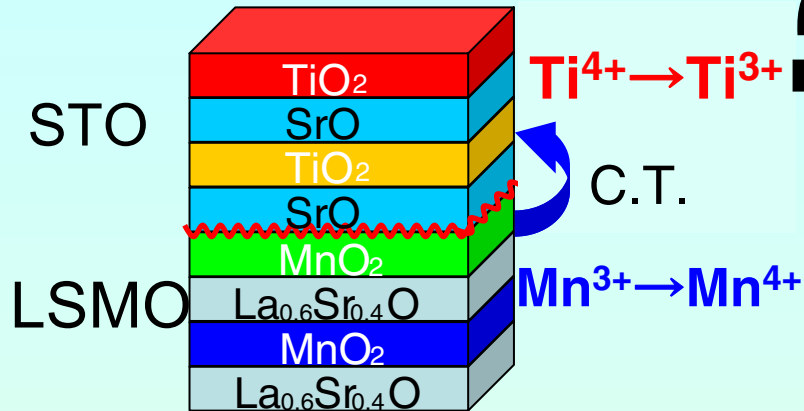


H. Wadati, H.K. et al., *Phys. Rev. B* 71, 035108 (05).

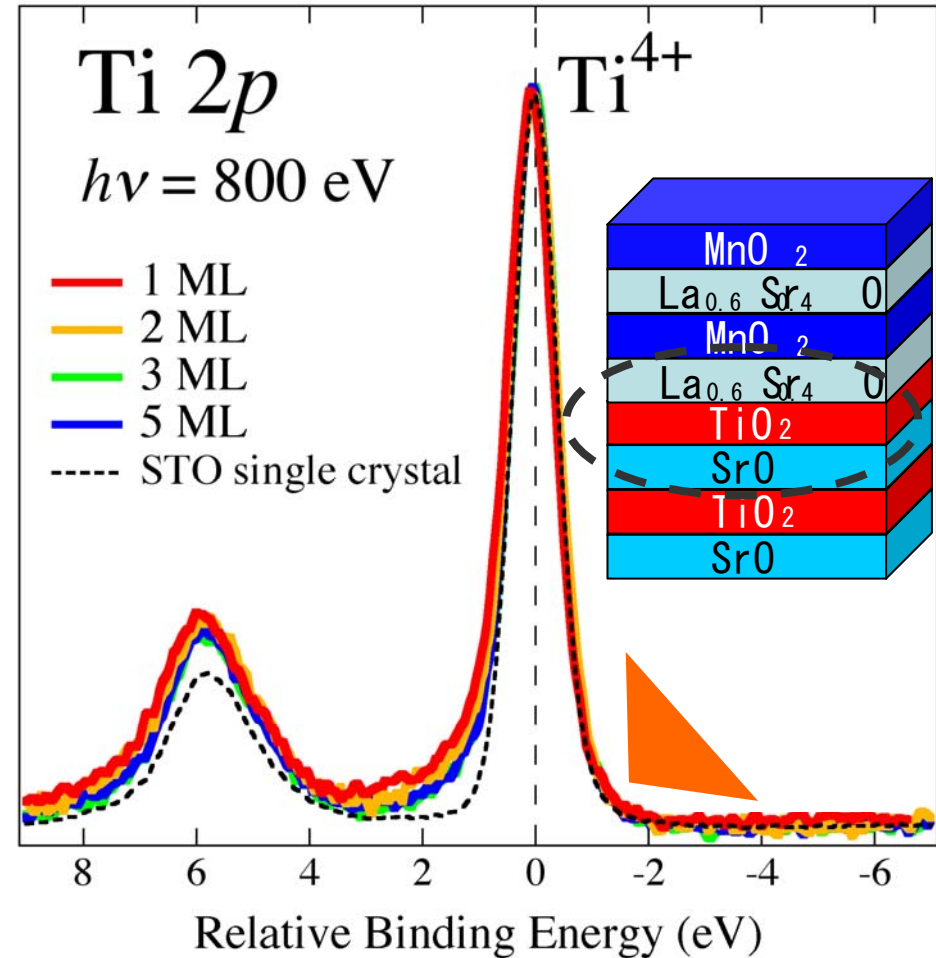
*Evidence of Charge Transfer
from LSMO to LSFO*

Ti 2p Core Level Spectra at Interfaces

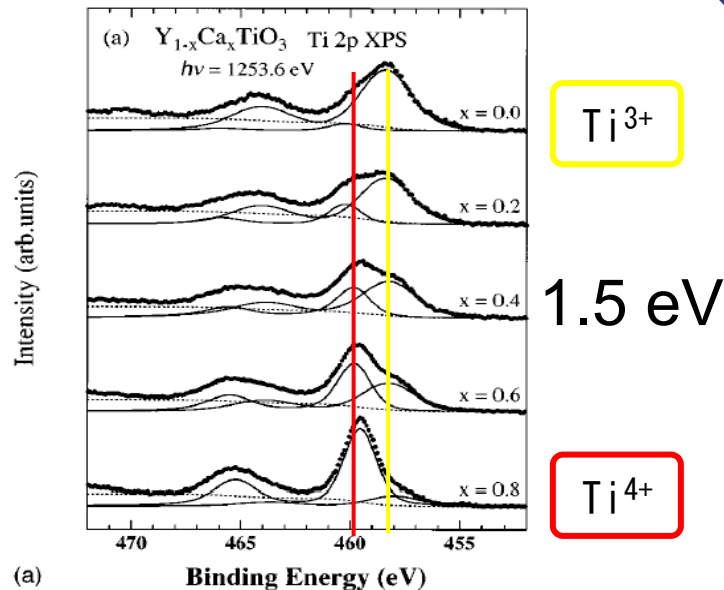
Charge Transfer ?



Intensity (arb. units)



H. Kumigashira *et al.*, Appl. Phys. Lett. **88**, 192504 ('06).

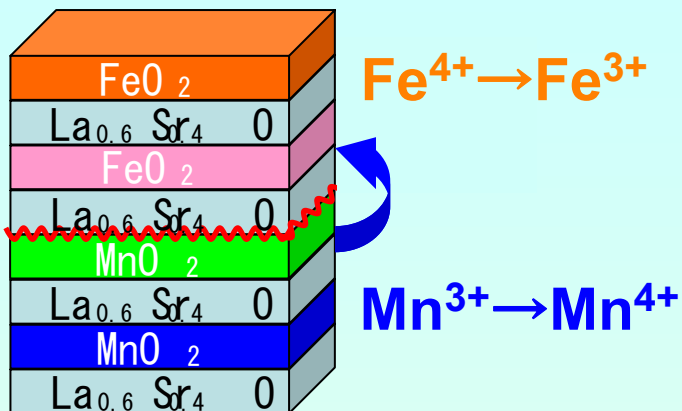


K. Morikawa *et al.*, Phys. Rev. B **54**, 5446 (1996).

Robust Ti⁴⁺ states

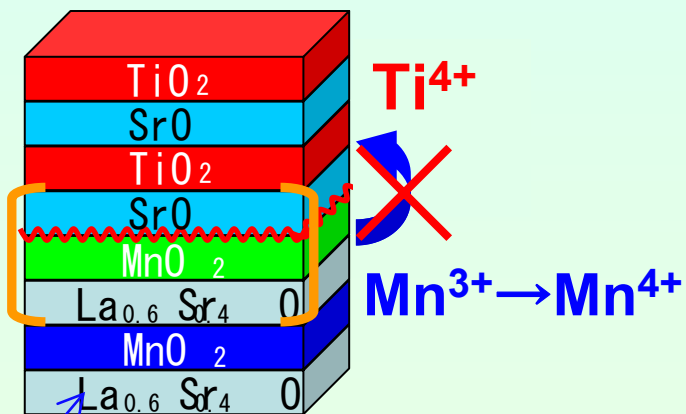
Origin of Charge Transfer

LSFO/LSMO

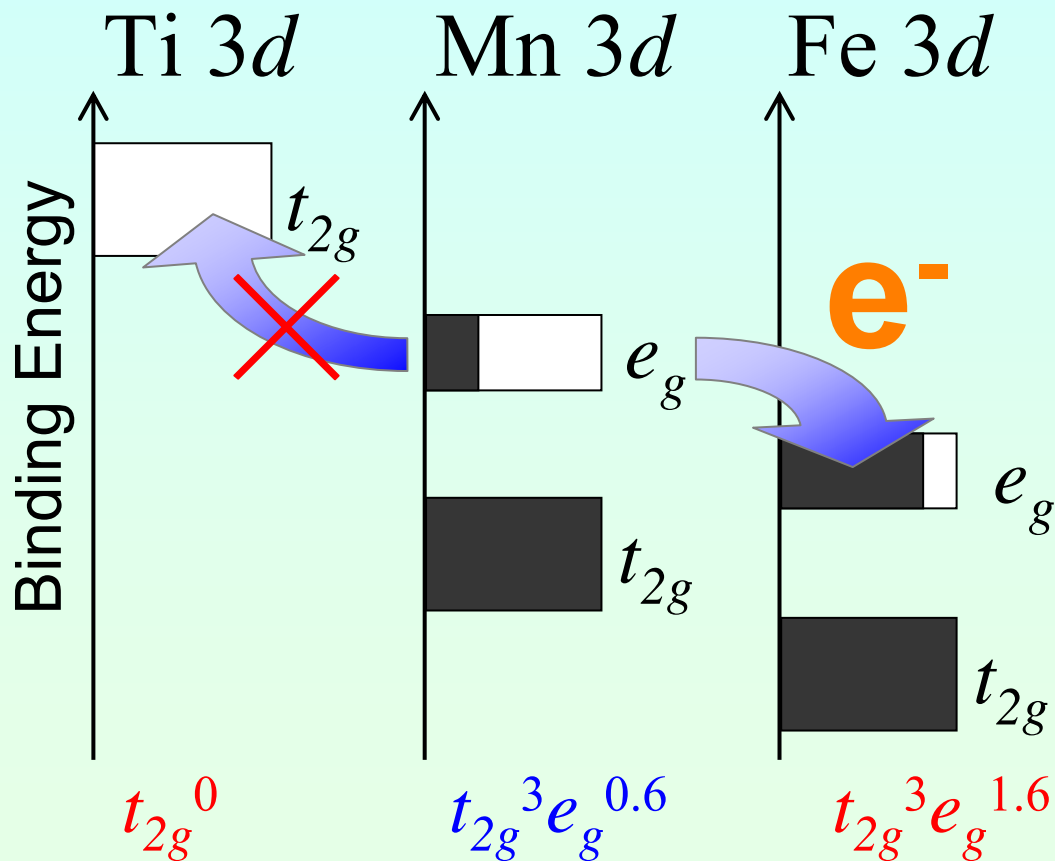


Difference of 3d levels among transition metals

STO/LSMO

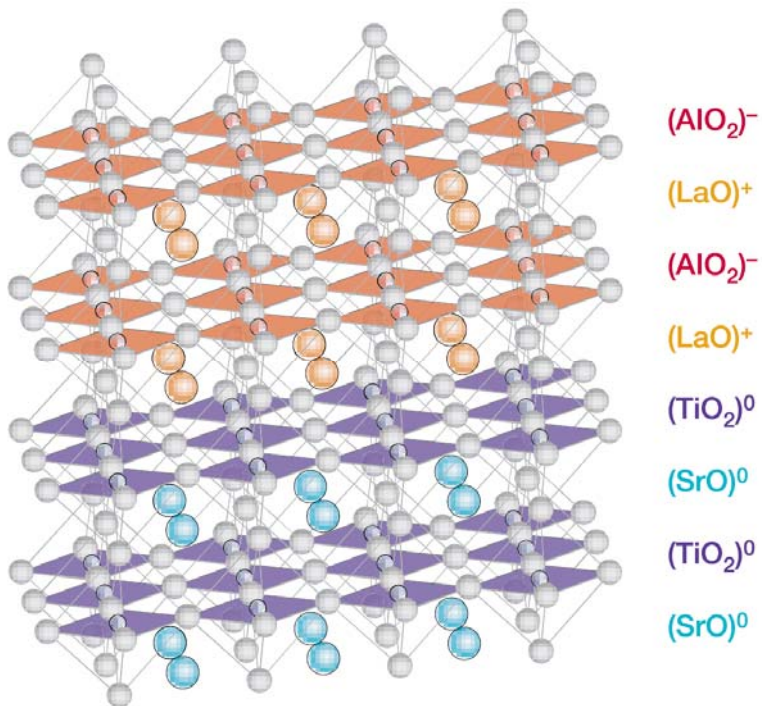


Electron-donor layer

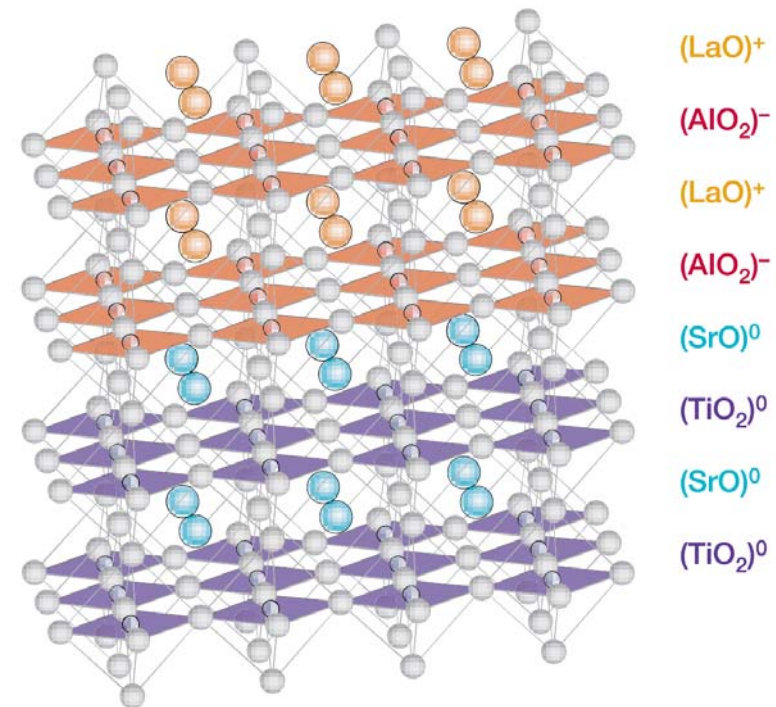


Charge redistribution at interface

Interfacial Electronic Structure of $\text{LaAlO}_3/\text{SrTiO}_3$ Heterojunctions



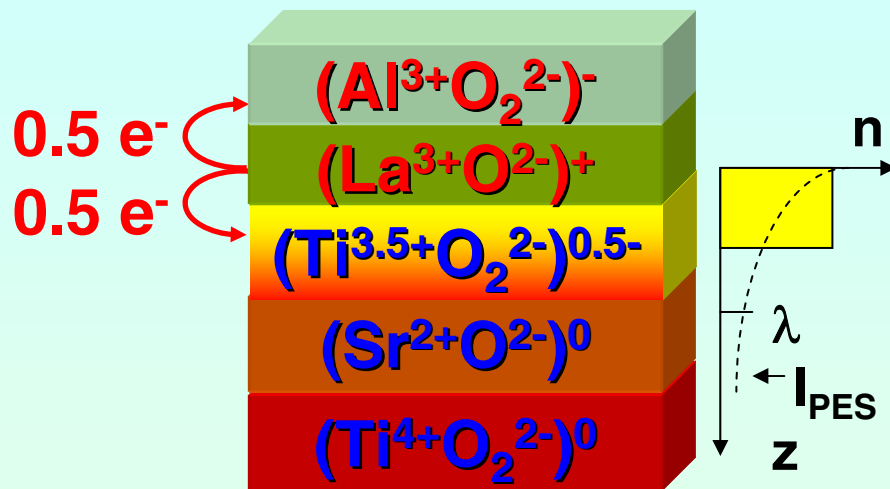
n-type (Metallic)



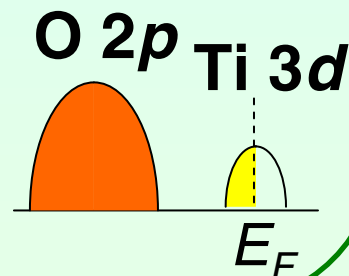
p-type (Insulating)

Origin of the Metallic Interface

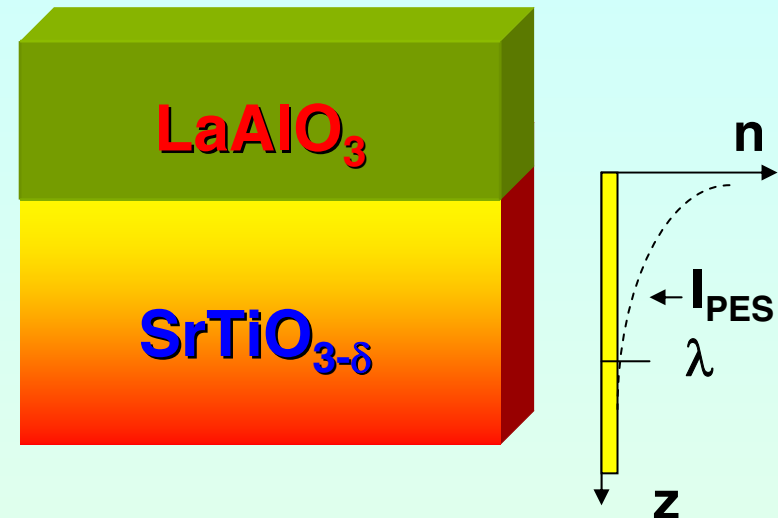
1. Charge Transfer



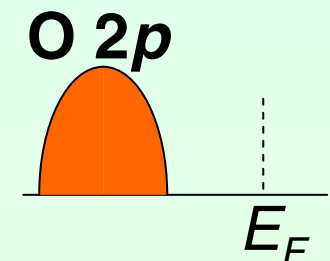
N. Nakagawa *et al.*,
Nature Mater. **5**, 204
(2006).



2. Oxygen Vacancies



W. Simons *et al.*,
Phys. Rev. Lett. **98**,
196802 (2007).

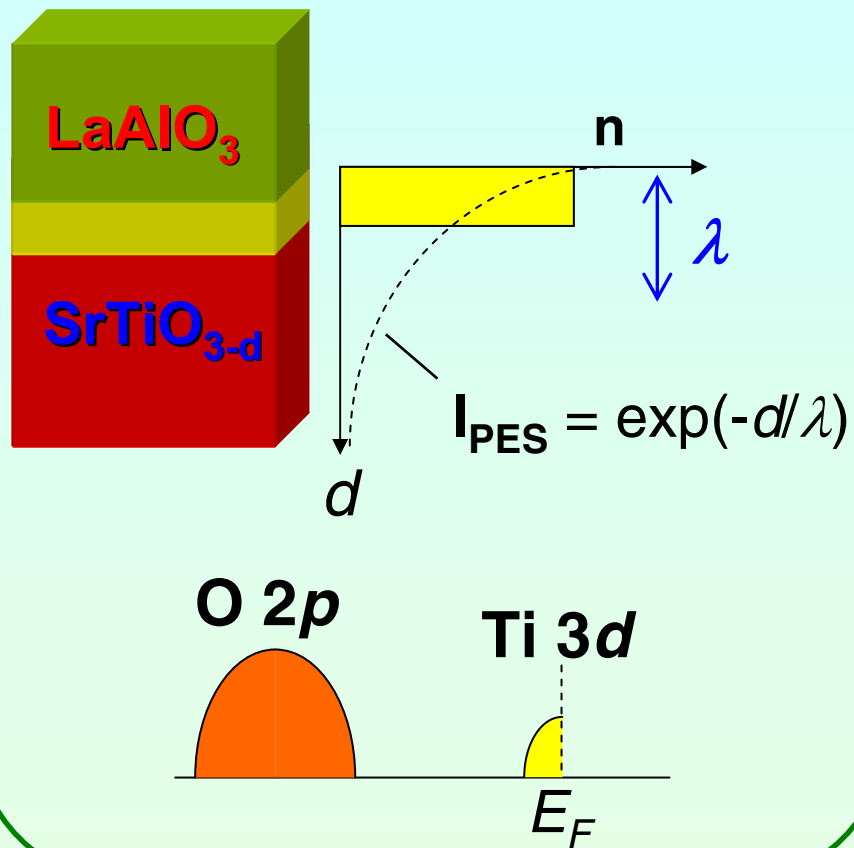


DOS at E_F is different between the two scenarios.

Detection Limit of PES Measurements

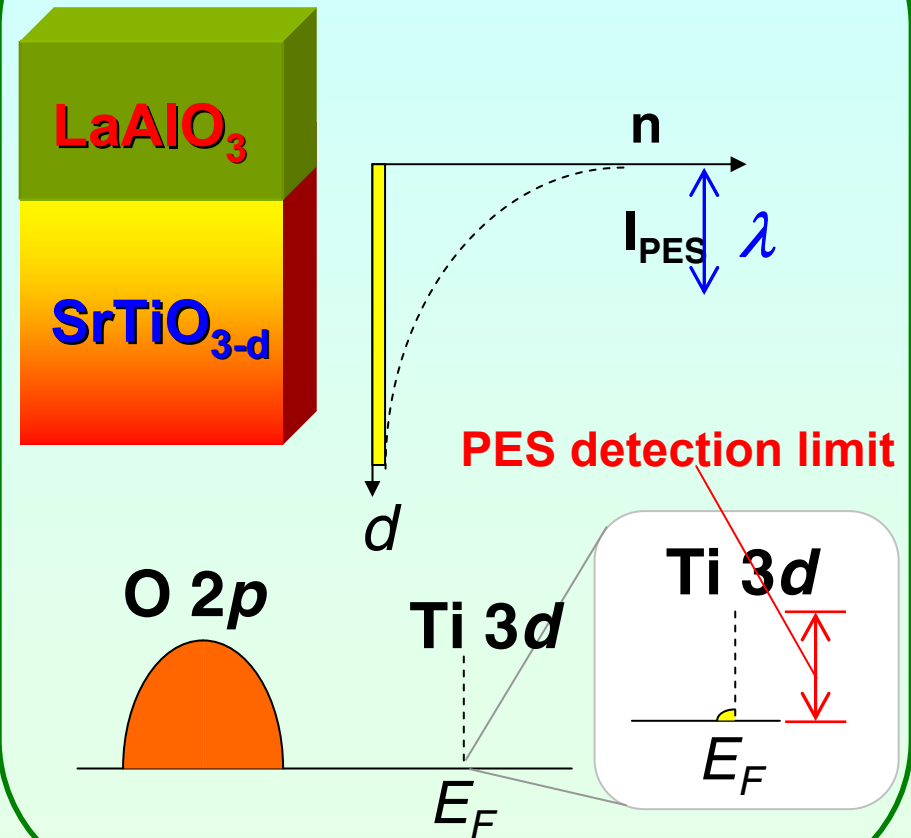
1. Charge Transfer

(Confinement@interface region)



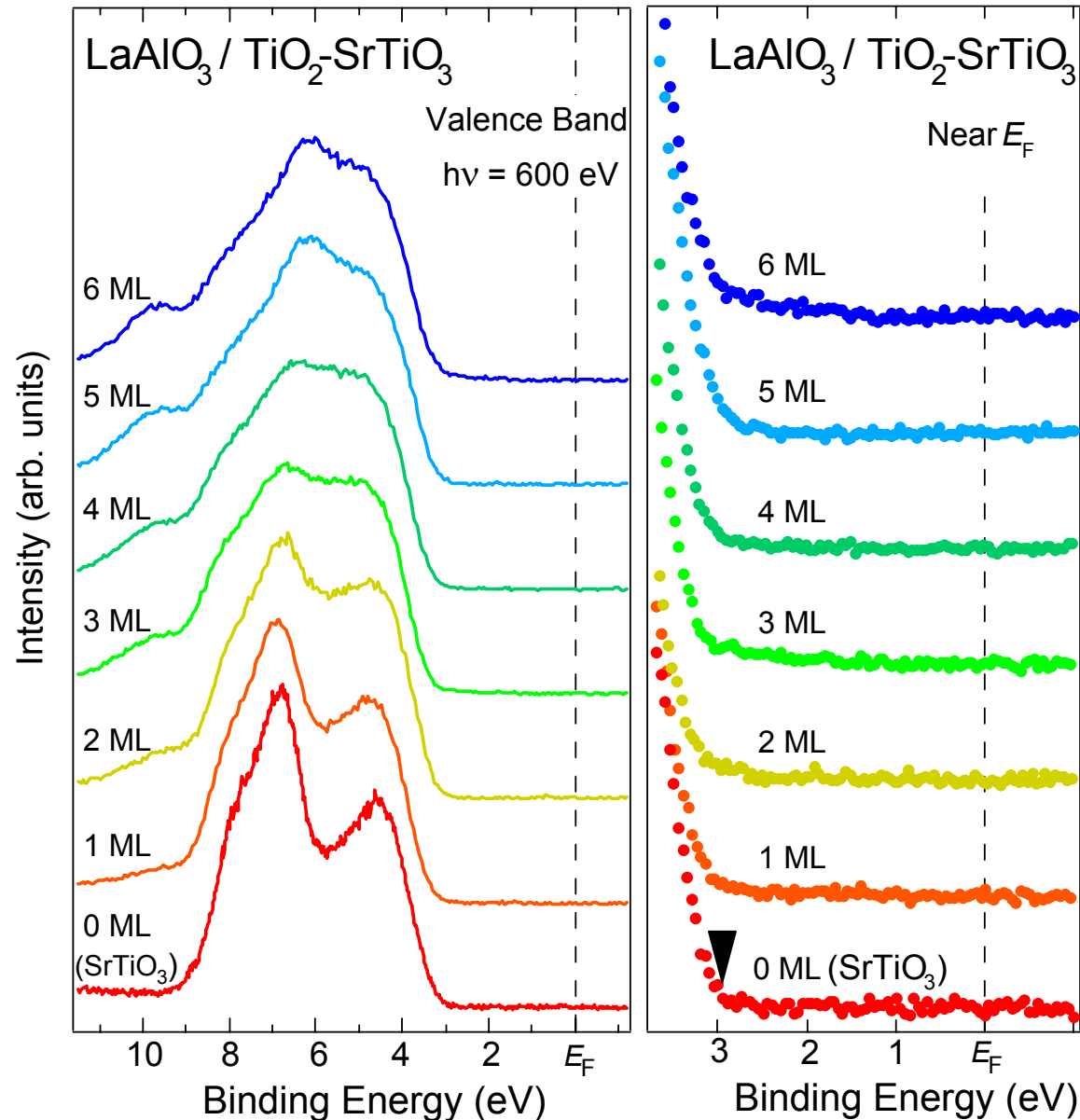
2. Oxygen Vacancies

(wide distribution over STO)

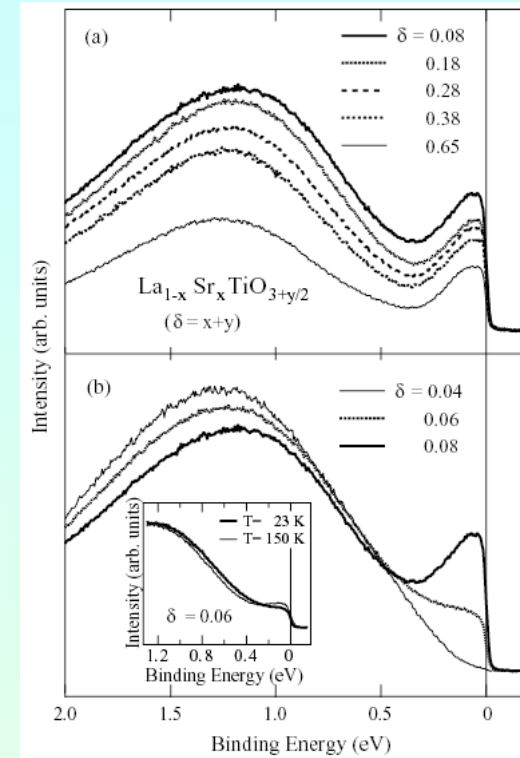


DOS at E_F is different between the two scenarios.

Valence Band Spectra of n-type Interfaces



PES spectra of LSTO

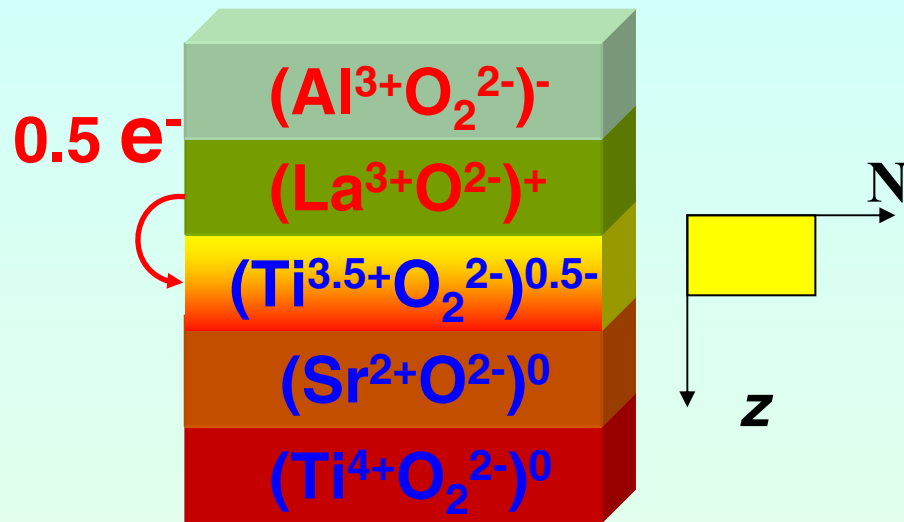


T. Yoshida *et al.*, Europhys. Lett. **59**, 258 ('02)

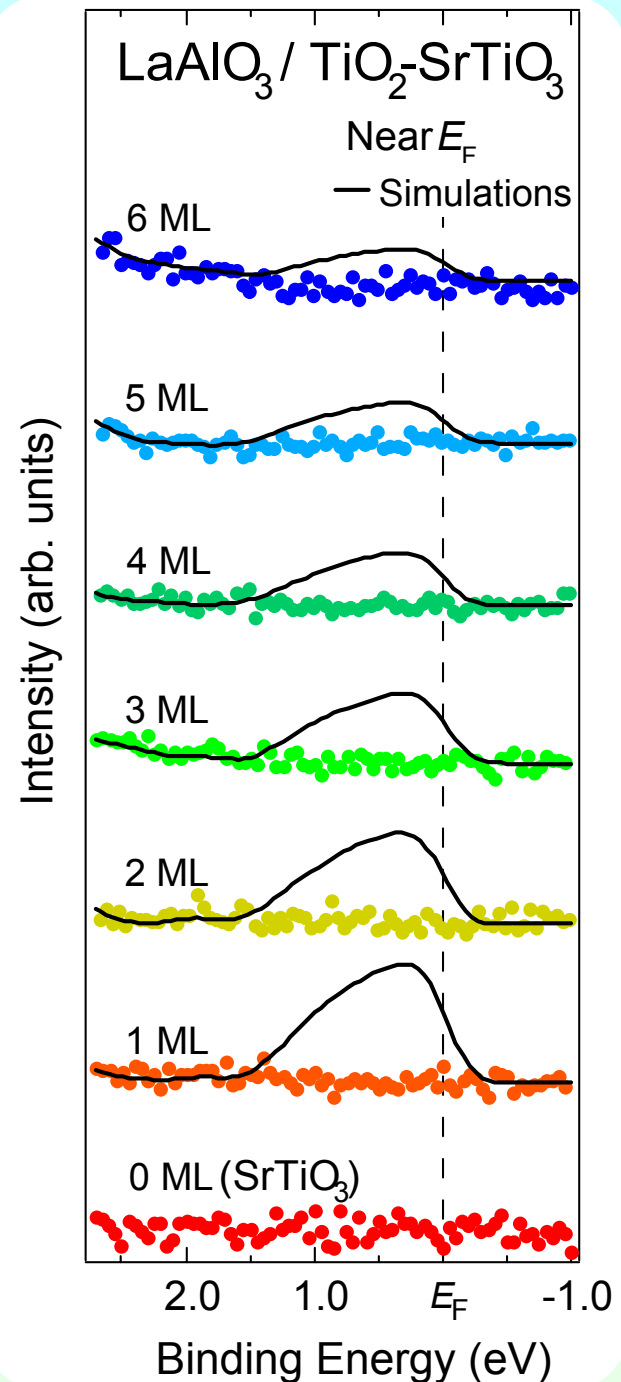
**No detectable
Ti 3d DOS at E_F**

Numerical Simulations

Confinement at the interface
(δ function)

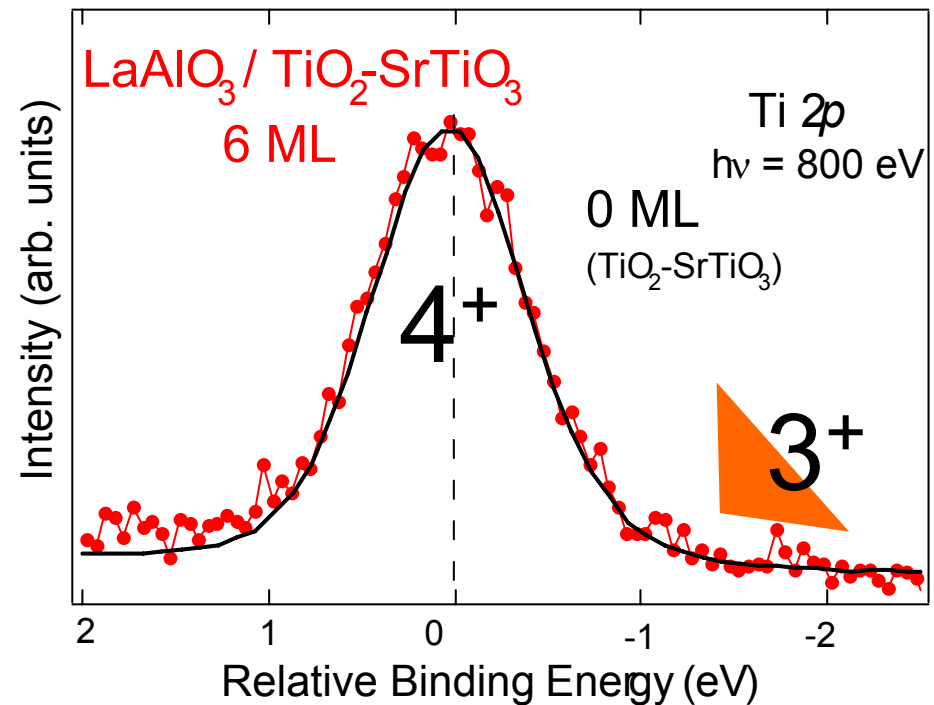
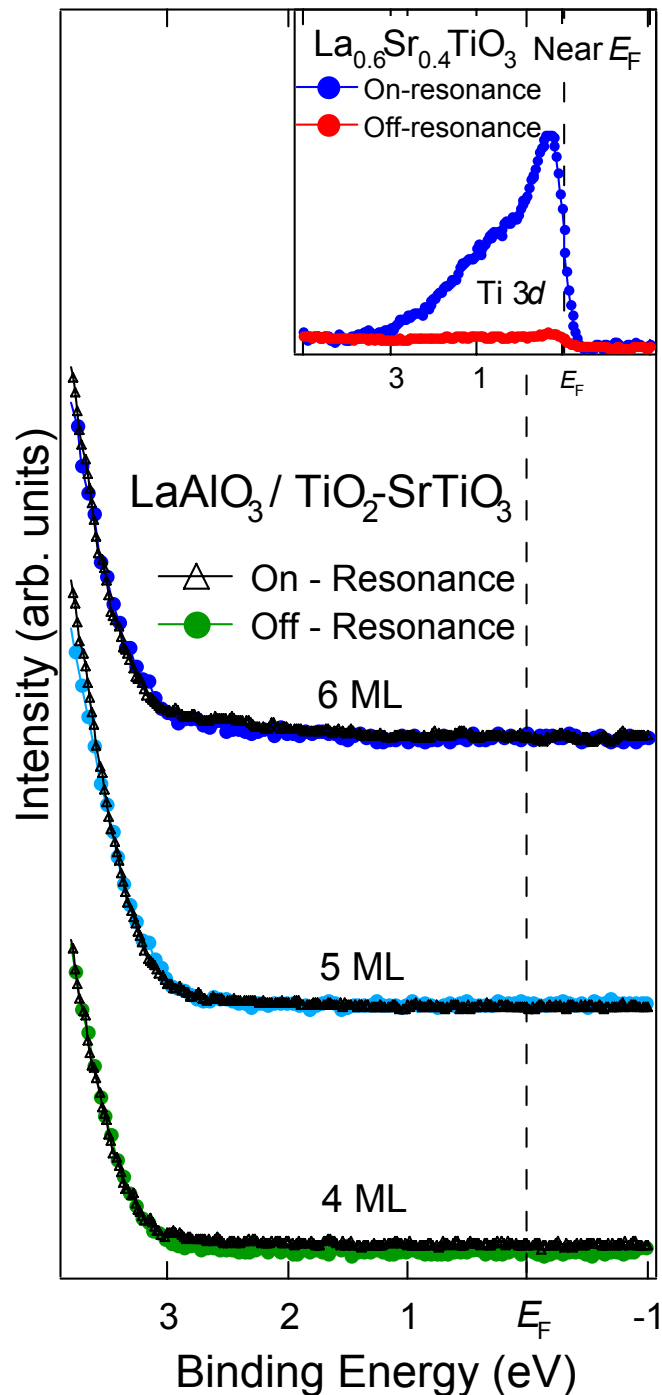


The 3d electrons only exist at TiO₂ layer adjacent to (LaO)⁺ layer.



Ti 2p → 3d Resonant Spectra

Chemical stabilities of Ti⁴⁺ states in TiO₂ irrespective of the neighboring donor LaO layer

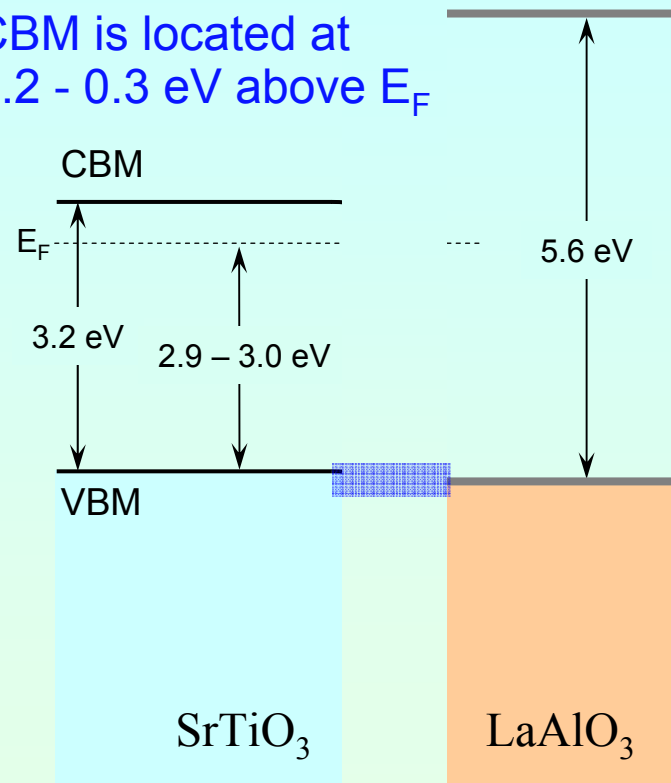


**Charge transfer doesn't occur
at the LAO/STO interface.**

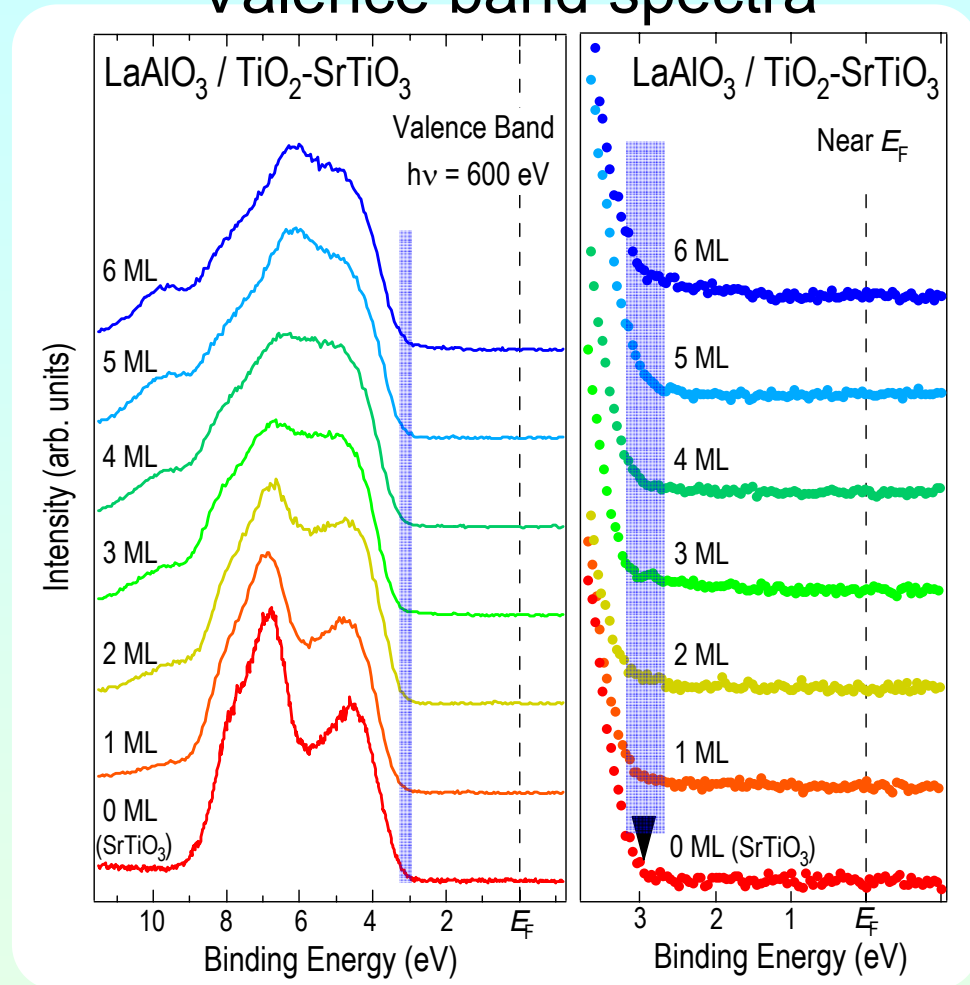
Band Diagram of the n-type Interface

$$E_g(\text{STO}) = 3.2 \text{ eV}$$
$$E_g(\text{LAO}) = 5.6 \text{ eV}$$

CBM is located at 0.2 - 0.3 eV above E_F



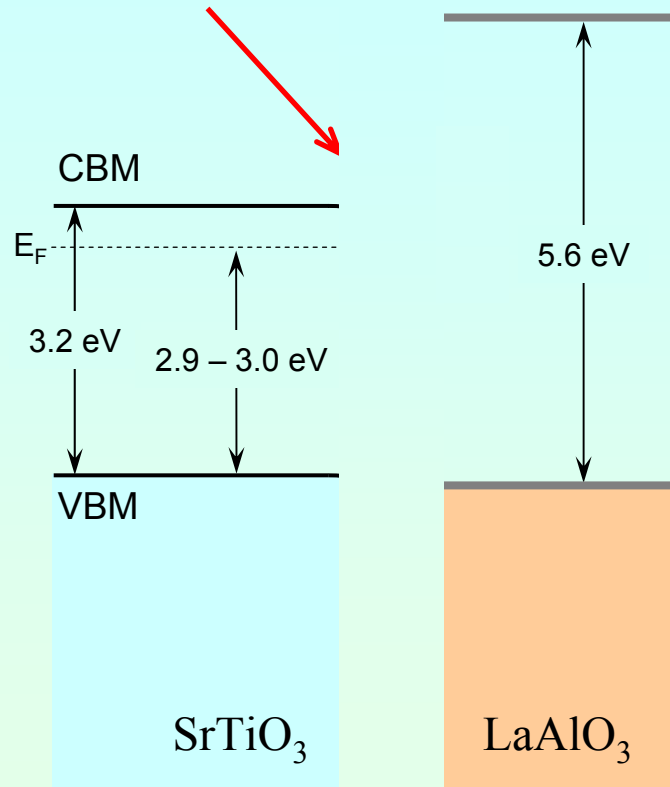
Valence band spectra



1. VBM of STO is located at 2.9-3.0 eV below E_F .
2. VBM is nearly continuous between STO and LAO.
(The leading edge of valence band structures is located at almost constant energy position.)

Band Diagram of the n-type Interface

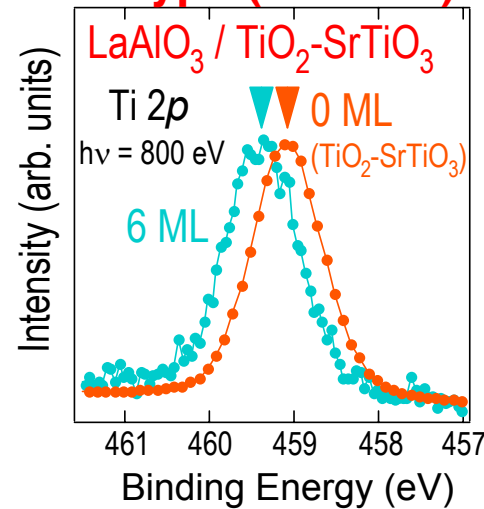
A Notched Stru



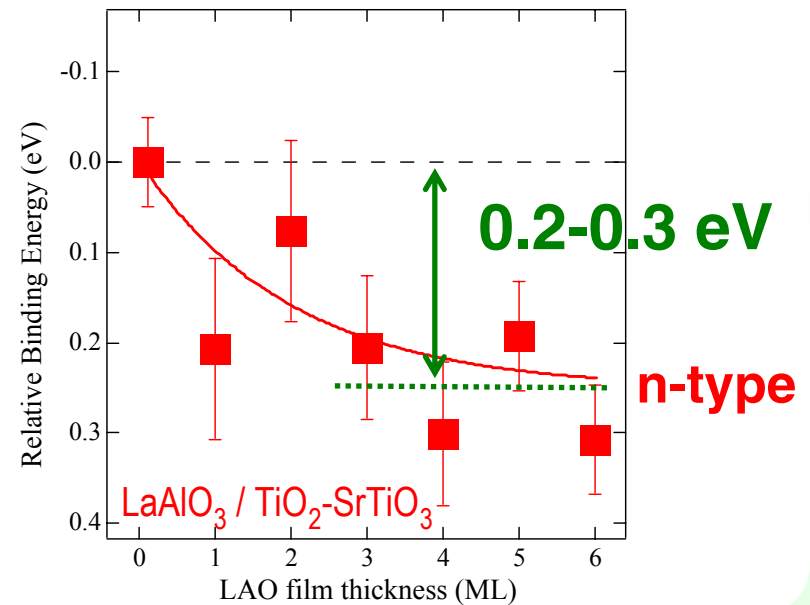
CBM in STO is nearly attained at E_F

Ti 2p core level spectra

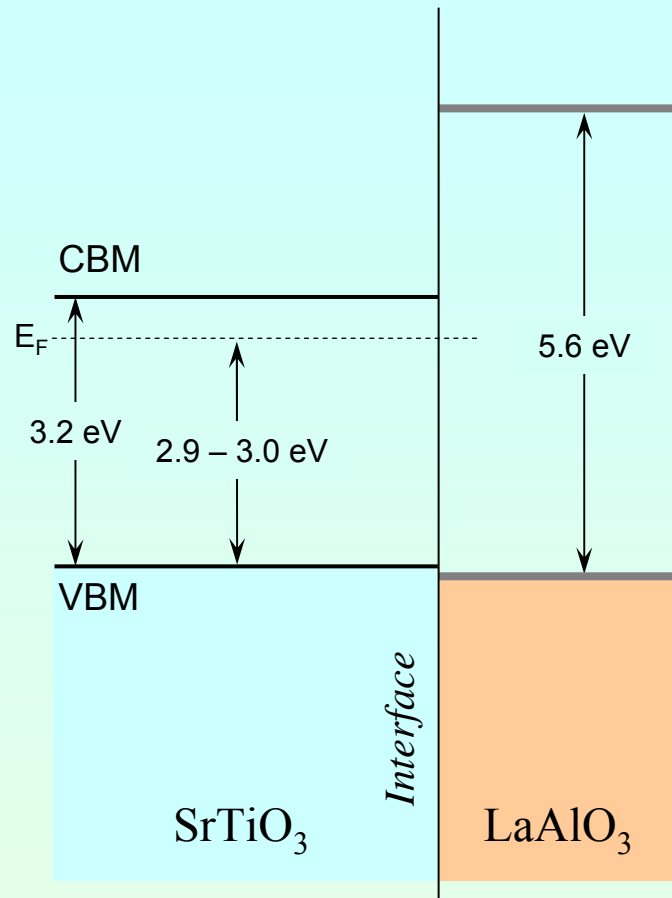
n-type (metallic)



Band bending of 0.2-0.3 eV to higher binding energy from STO to the metallic interface.



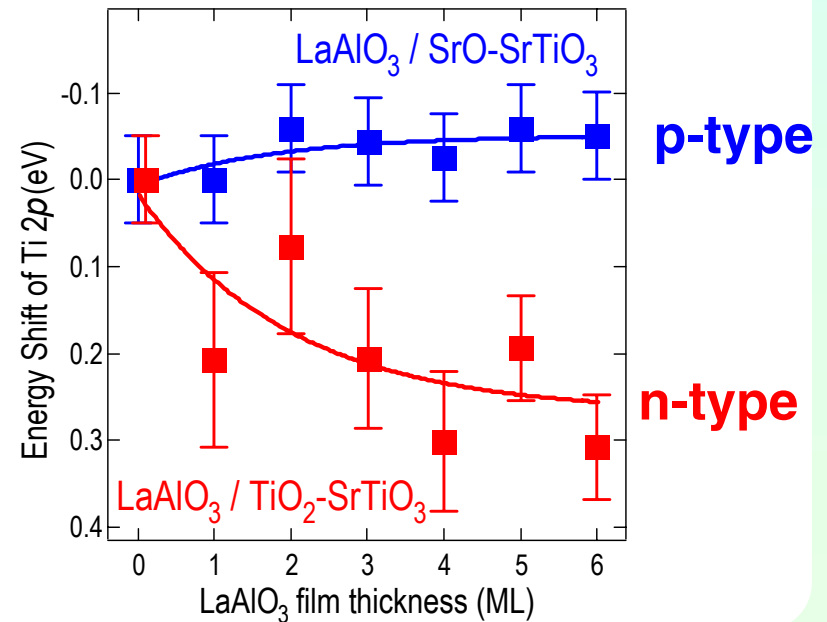
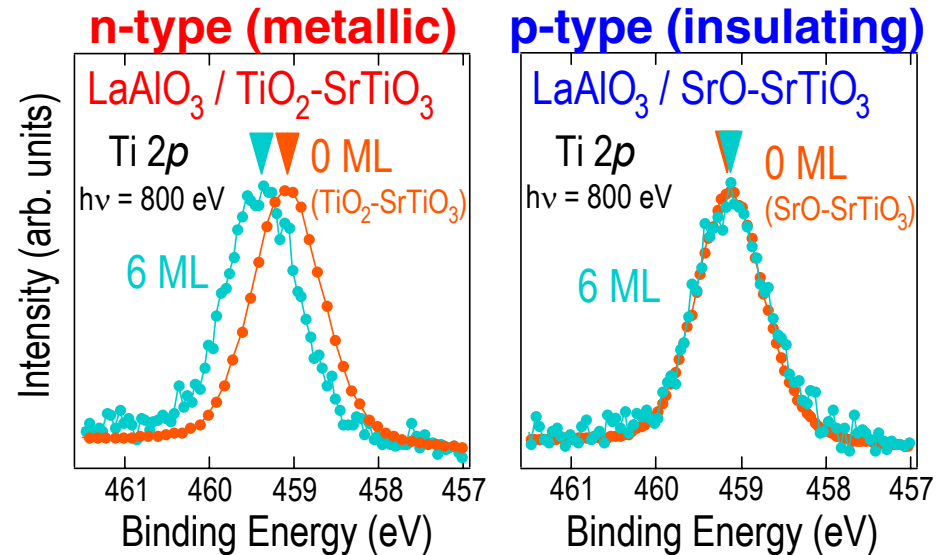
Band Diagram of LAO/STO Interface



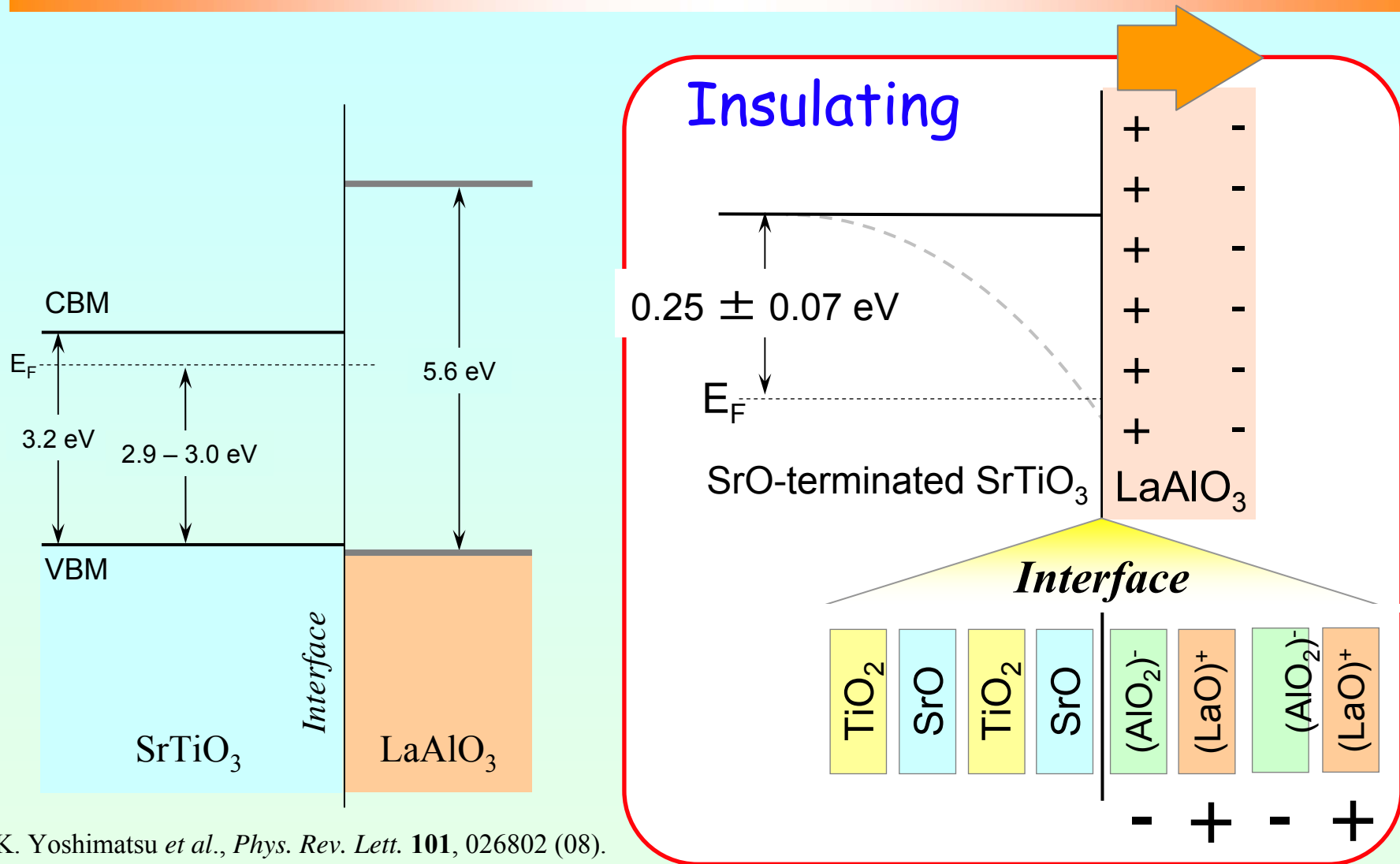
No peak shift at the p-type interface

Band bending on the STO side is responsible for the metallic states.

Ti 2p core level spectra



Origin of M-I Transition by Inserting SrO



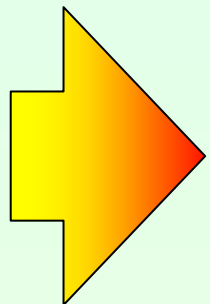
The notched structure disappears by inserting SrO atomic layer between LAO and STO.

Summary in LAO/STO Interface

We determined the band diagrams of LAO/STO heterojunctions by using *in situ* photoemission spectroscopy.

We have found

1. There is no detectable Ti 3d DOS at E_F expected from charge transfer through the interface.
2. Owing to the band discontinuity of LAO/STO, a notched structure is formed at the metallic interface.
3. The structure, however, is absent at the insulating interface.



The metallic states at the interface between band insulators LAO/STO originate not from charge transfer through the interface on a short-range scale but from the accumulation of carrier on a long-range scale.

Conclusion

1. レーザーMBE法(MPLD法)の発展により、原子レベルで構造を制御した酸化物薄膜・超格子の作製が可能となった。
2. 強相関酸化物においても、今後、物質設計の自由度が高く、物性を自由自在に制御可能な「超構造」を用いた研究が主流になる。
3. 放射光を用いた分光法は、酸化物薄膜・超格子研究の強力な牽引力になる。

